



# Shaping the Future of Electron Microscopy: The TEAM Microscope

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94720, USA**

## **TEAM is a collaborative Project**

**T**ransmission **E**lectron **A**bserration-corrected **M**icroscope

Lawrence Berkeley National Laboratory

Argonne National Laboratory

Brookhaven National Laboratory

Frederick-Seitz Materials Research Lab, UIUC

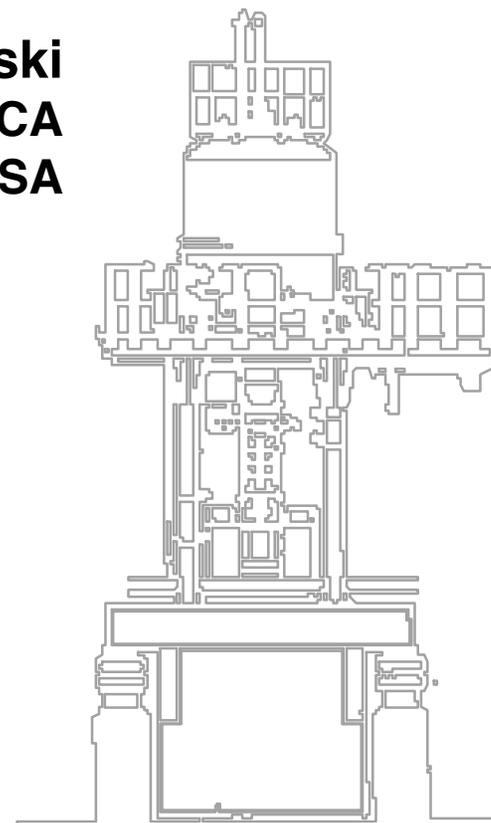
Oak Ridge National Laboratory

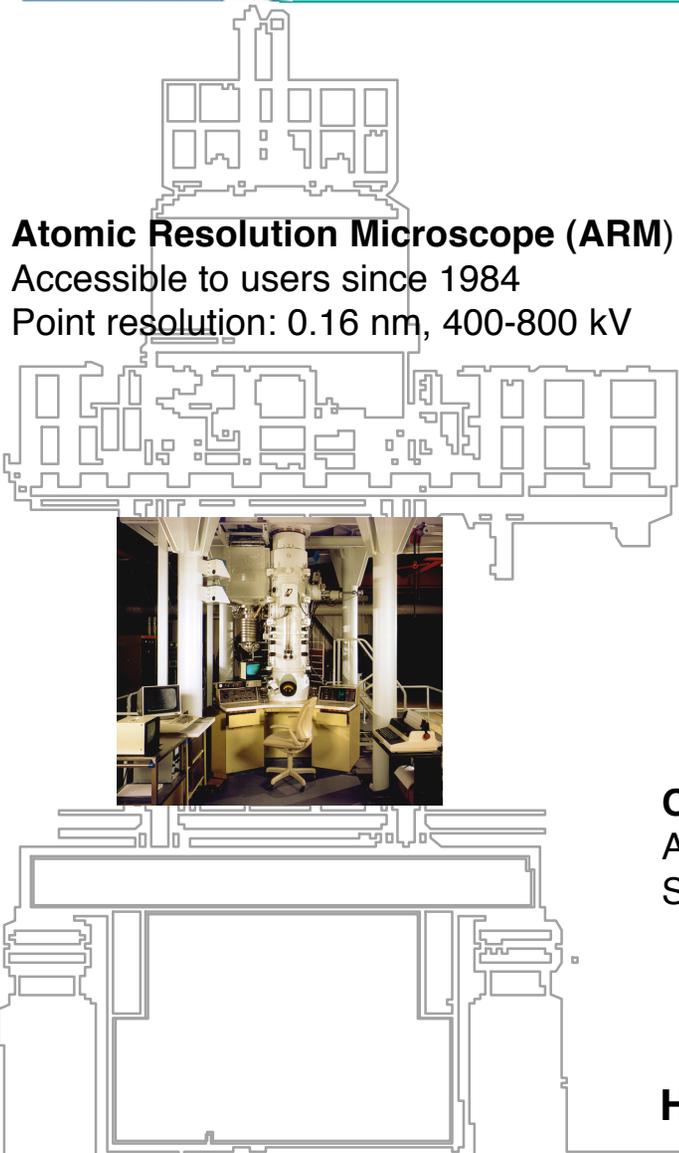
**Industry: FEI Company, CEOS**

**NCEM: A.Schmid, A. Minor,**

**U. Dahmen (Head), P. Denes (Manager)**

**Supported by the US Department of Energy**





## Atomic Resolution Microscope (ARM)

Accessible to users since 1984

Point resolution: 0.16 nm, 400-800 kV



## One Ångstrom Microscope (OAM)

Accessible to users since 1999

Spatial resolution: 0.08 nm, 300 kV



**Highest TEM resolution today**



## STEM / HRTEM : Tecnai G<sup>2</sup>

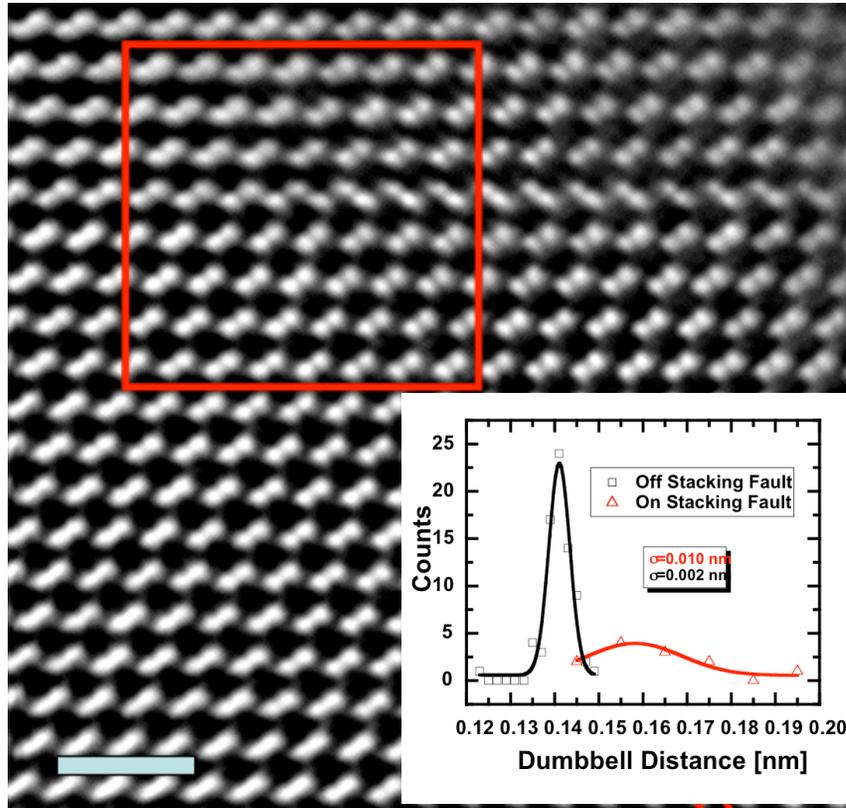
Accessible since 2003, 200 kV

Spatial resolution: 1.2 Ångstrom

Energy resolution: 150 meV

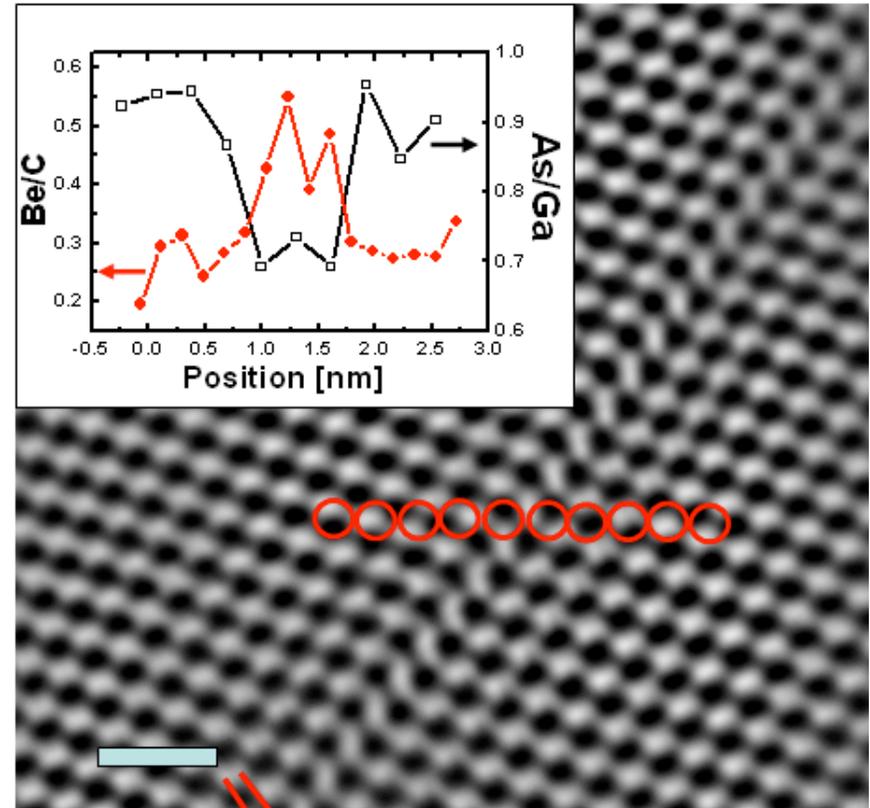
# A STEM / TEM Study: First full analysis of a single dislocation, GaAs:Be

## Electron Exit Wave Reconstruction Position determination



0.14nm

## STEM (SuperSTEM, UK) Spectroscopy

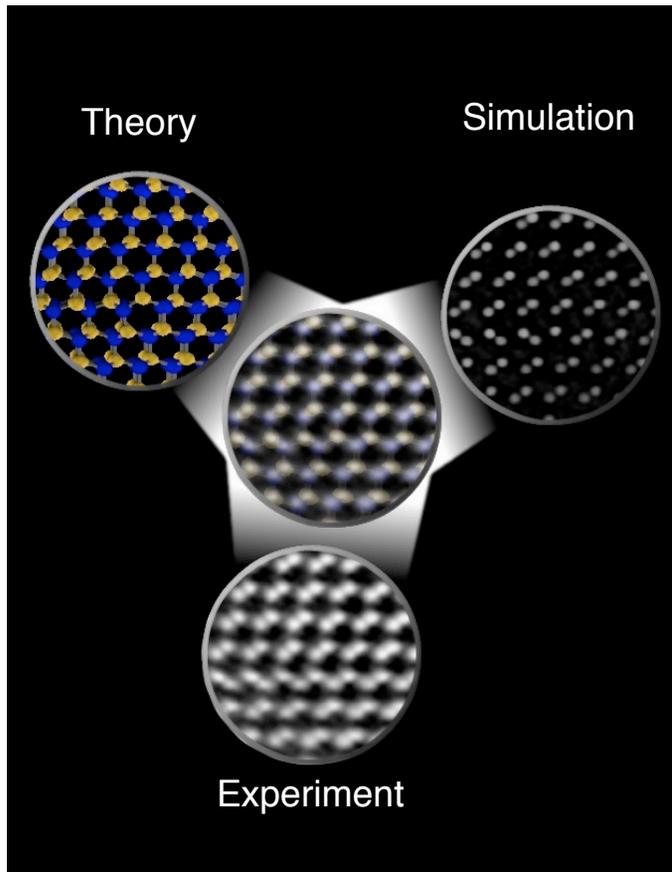


0.14nm

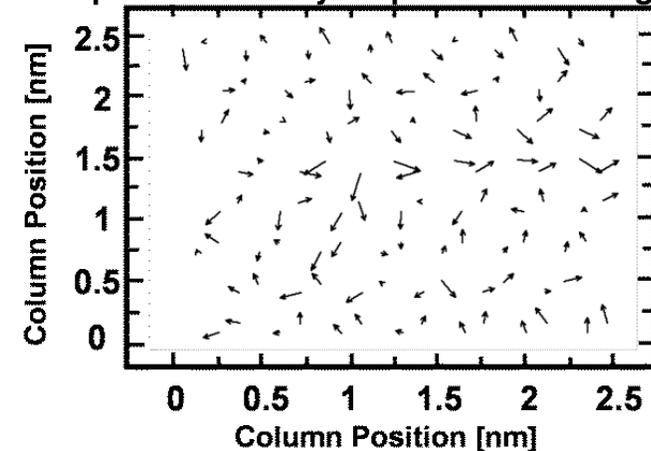
- Column positions can be determined to pm precision
- Compositional changes can be revealed on single atom columns

# Combining Experiments & Simulations & Theory

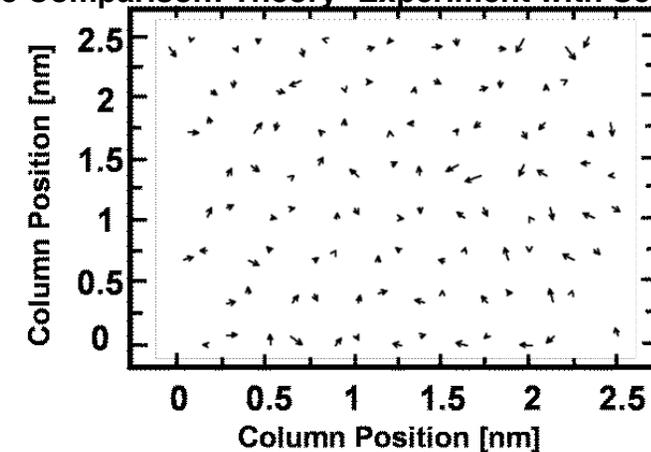
## Ga terminated 30° partial dislocation in GaAs: Be



Quantitative Comparison: Theory -Experiment w/o Segregation



Quantitative Comparison: Theory -Experiment with Segregation

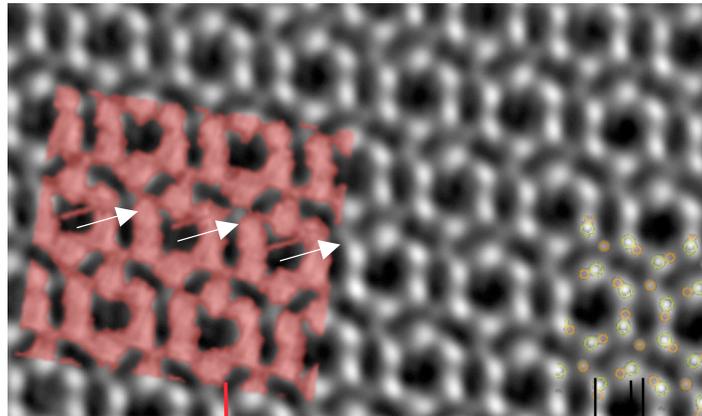


- Theory and experiment can be merged on the same scale (~ 4000 atoms)
- Microscopy can guide theory

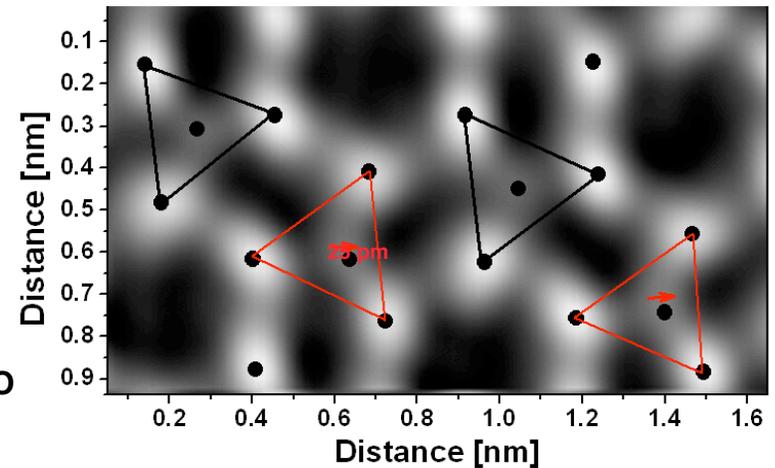
X.Xu, S.P. Beckmann, P. Specht, E.R. Weber, D.C. Chrzan, R.P. Erni, I. Aslan, N. Brwoning, A. Bleloch, C. Kisielowski, PRL 95, 2005, 145501

# Why Engage in New Developments

OAM reconstructed phase image (2005):



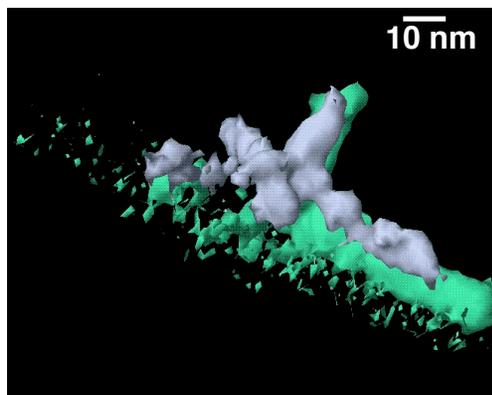
$\text{Si}_3\text{N}_4$ : Al, O



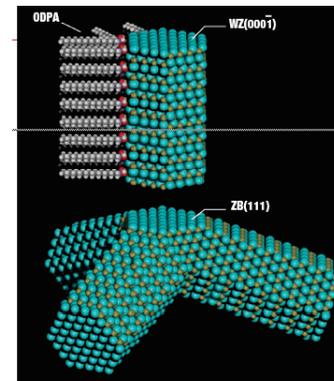
ARM lattice image (1988):

$\text{N}_2$   $\text{N}_1$  Si

- Projection of 3D object in 2D image
- Intensities are not interpreted

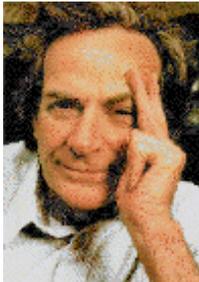


•3D with atomic resolution



# The Feynman Challenge

“It would be very easy to make an analysis of any complicated chemical substance; all one would have to do would be to look at it and see where the atoms are. The only trouble is that the electron microscope is one hundred times too poor ... I put this out as a challenge: **Is there no way to make the electron microscope more powerful?**”



– Richard P. Feynman, 1959,  
“There’s Plenty of Room at the Bottom”

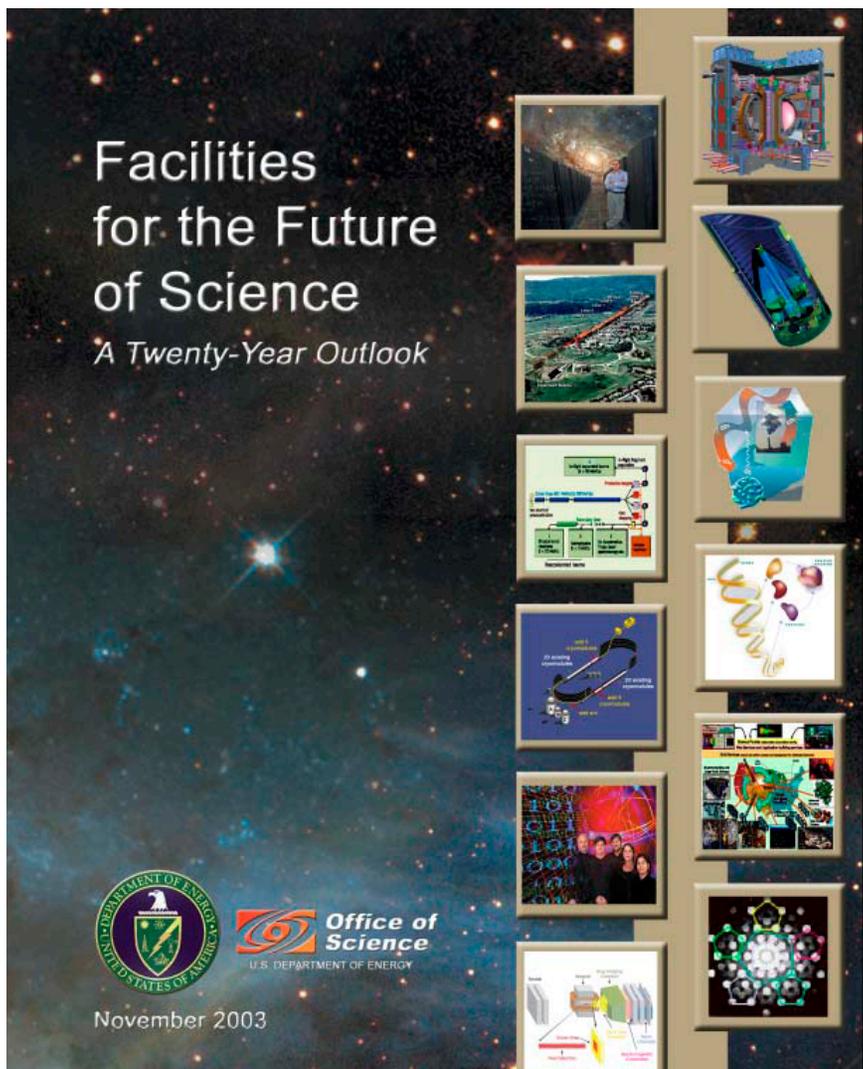
## TEAM aims at meeting the Feynman challenge

There are numerous other advantages that are not mentioned



# The DoE is Committed to Support TEAM

NCEM —



**TEAM is a DoE Project**

**High visibility, near term goal**



**Platform: TITAN**

Access: 2007 - 2009

Spatial resolution:

0.5 Ångstrom STEM/TEM

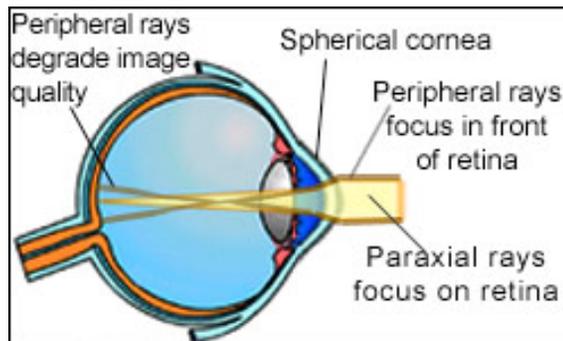
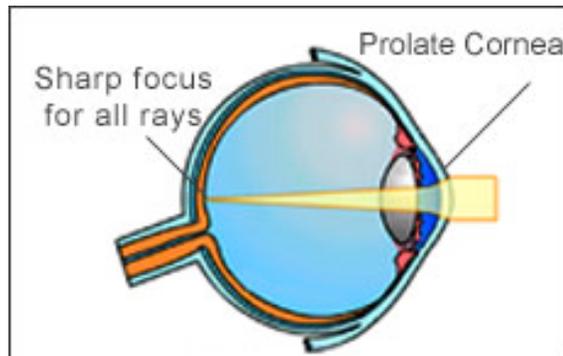
Energy resolution <100 meV

80 - 300 kV

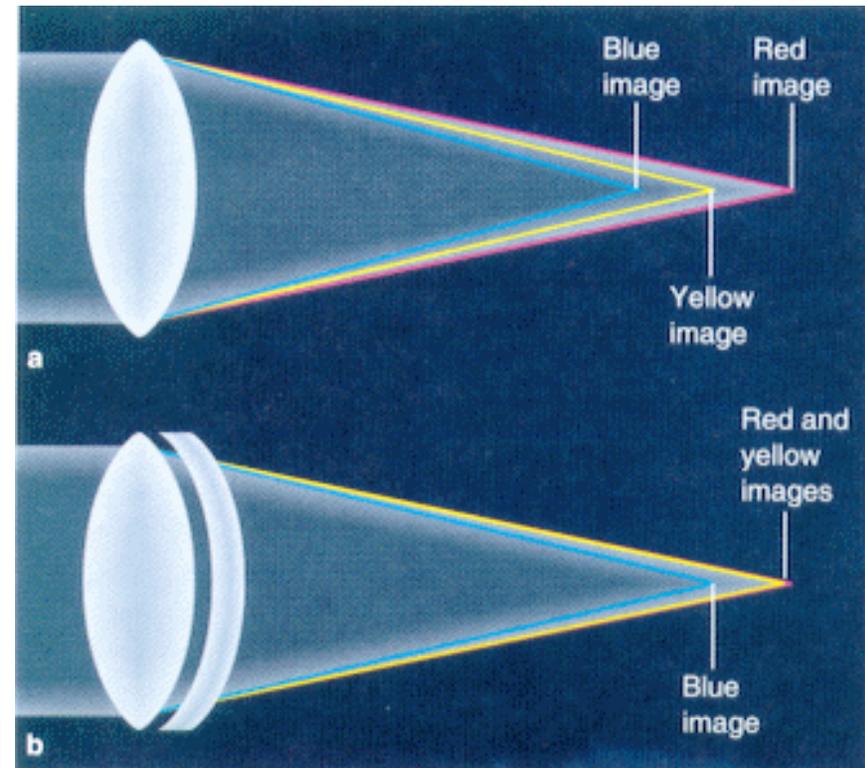
**Aberration corrected**

# Well Known Aberrations

## Spherical aberration



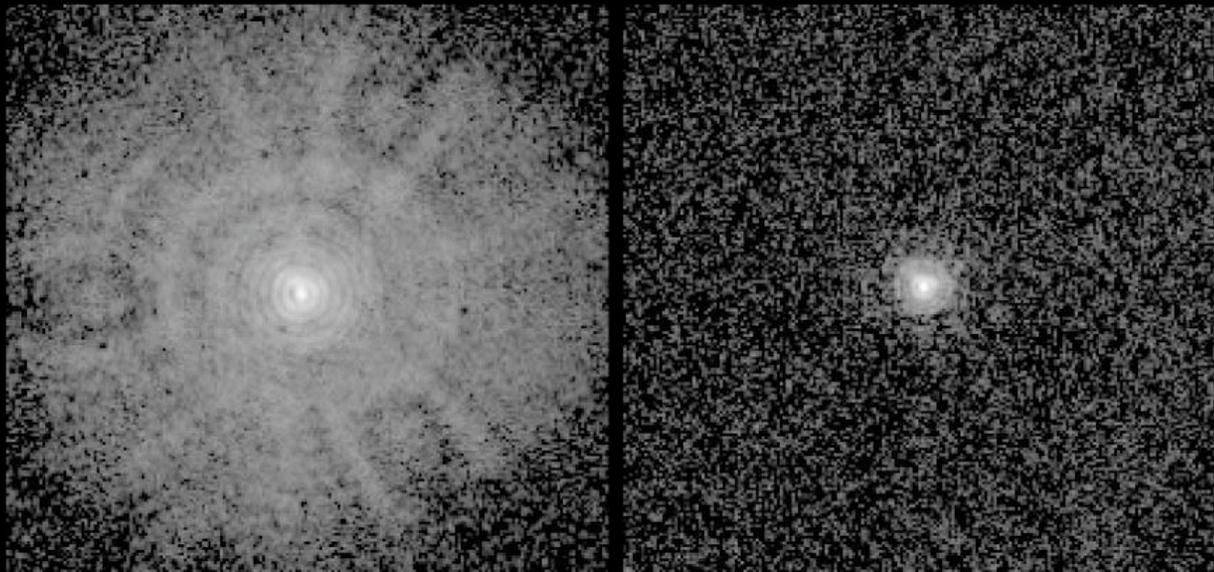
## Chromatic aberration



...and how they are corrected

# A Famous Example

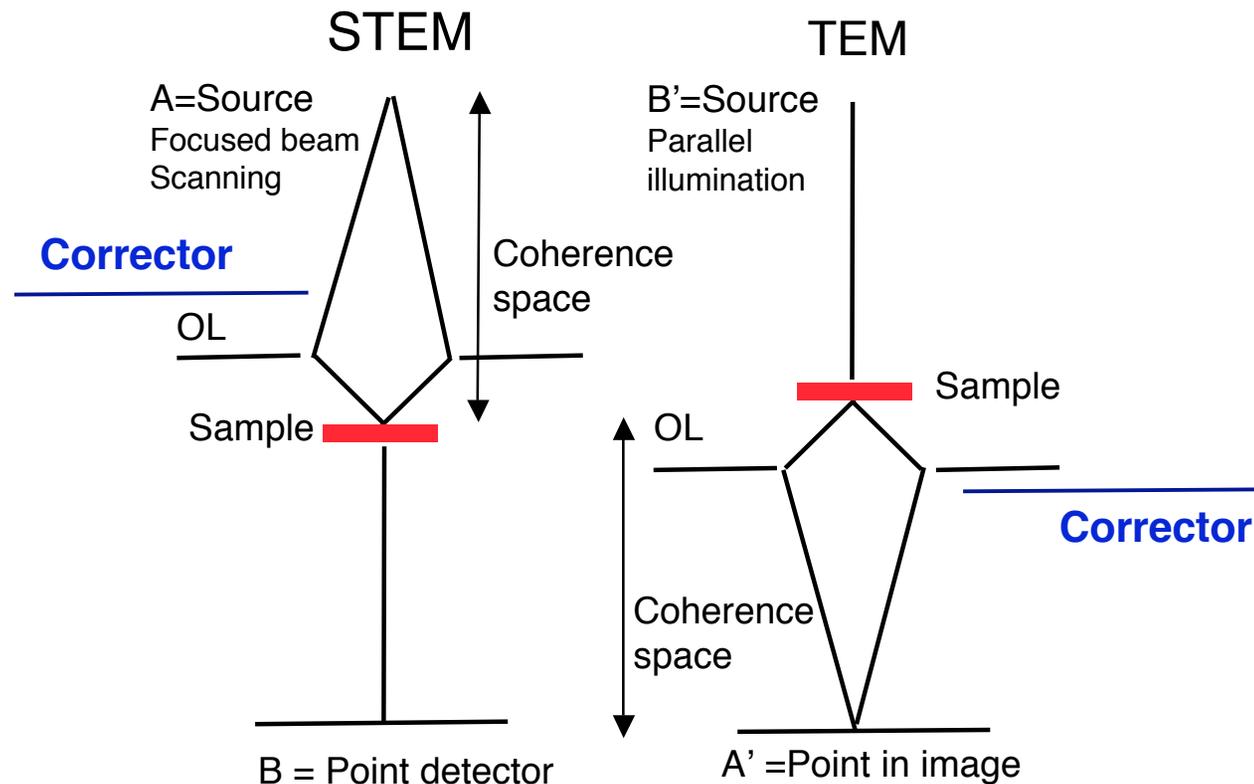
## HUBBLE SPACE TELESCOPE FAINT OBJECT CAMERA COMPARATIVE VIEWS OF A STAR



BEFORE COSTAR

AFTER COSTAR

# STEM & TEM

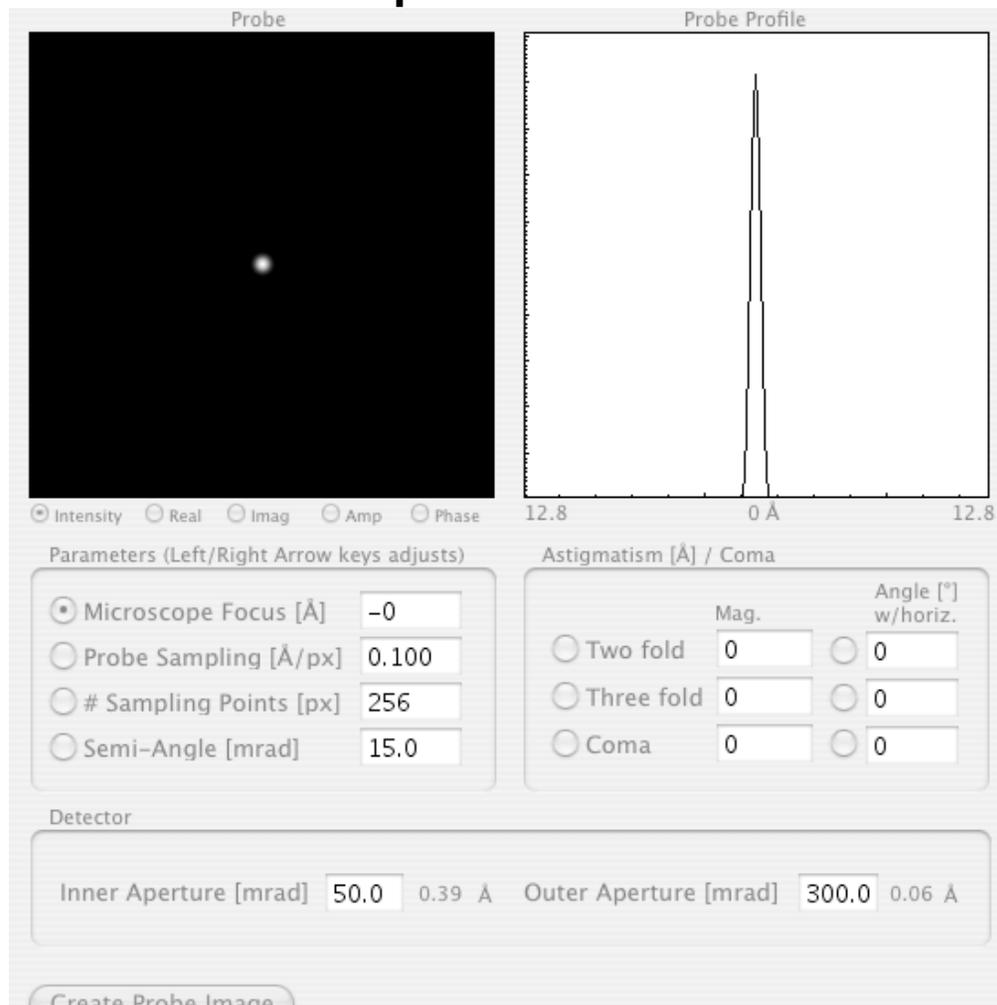
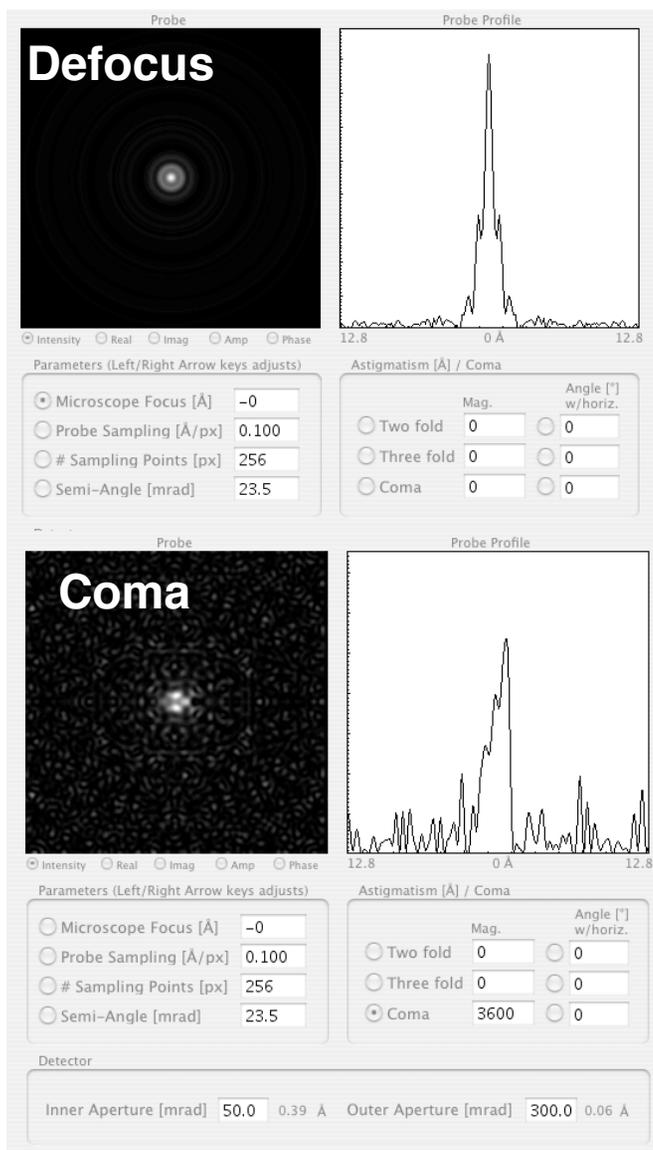


- **Principle of reciprocity:**  
Electron trajectories and elastic scattering processes have time reversal symmetry. J.M Cowley (APL 15, 1969,58)
- **TEM and STEM can perform identical tasks**  
Differences relate to S/N ratios & experimental set up  
Differences MATTER!

# Lens Aberrations

## STEM: Probe

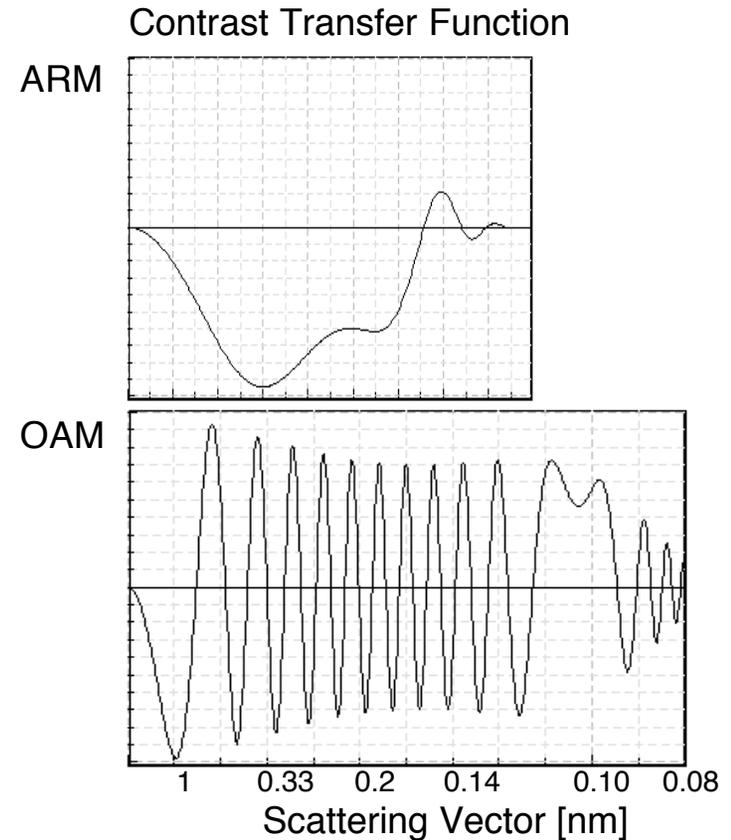
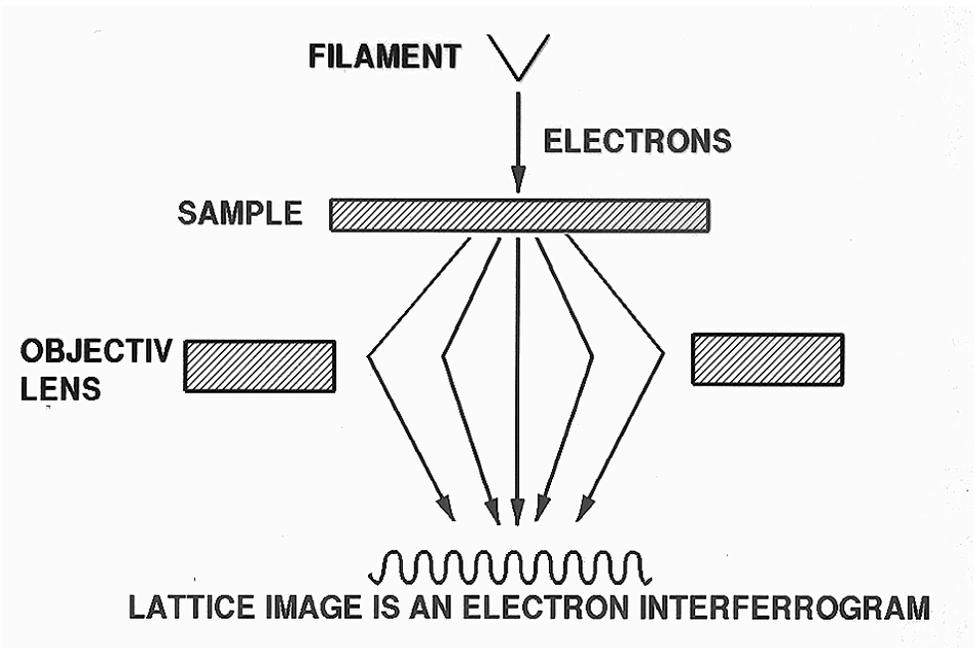
### Cs corrected probe



**In STEM aberration can only be corrected by hardware**

# Principle of HRTEM

## Image formation



- We record Intensities:  $I \sim \Psi\Psi^*$

Loss of phase information

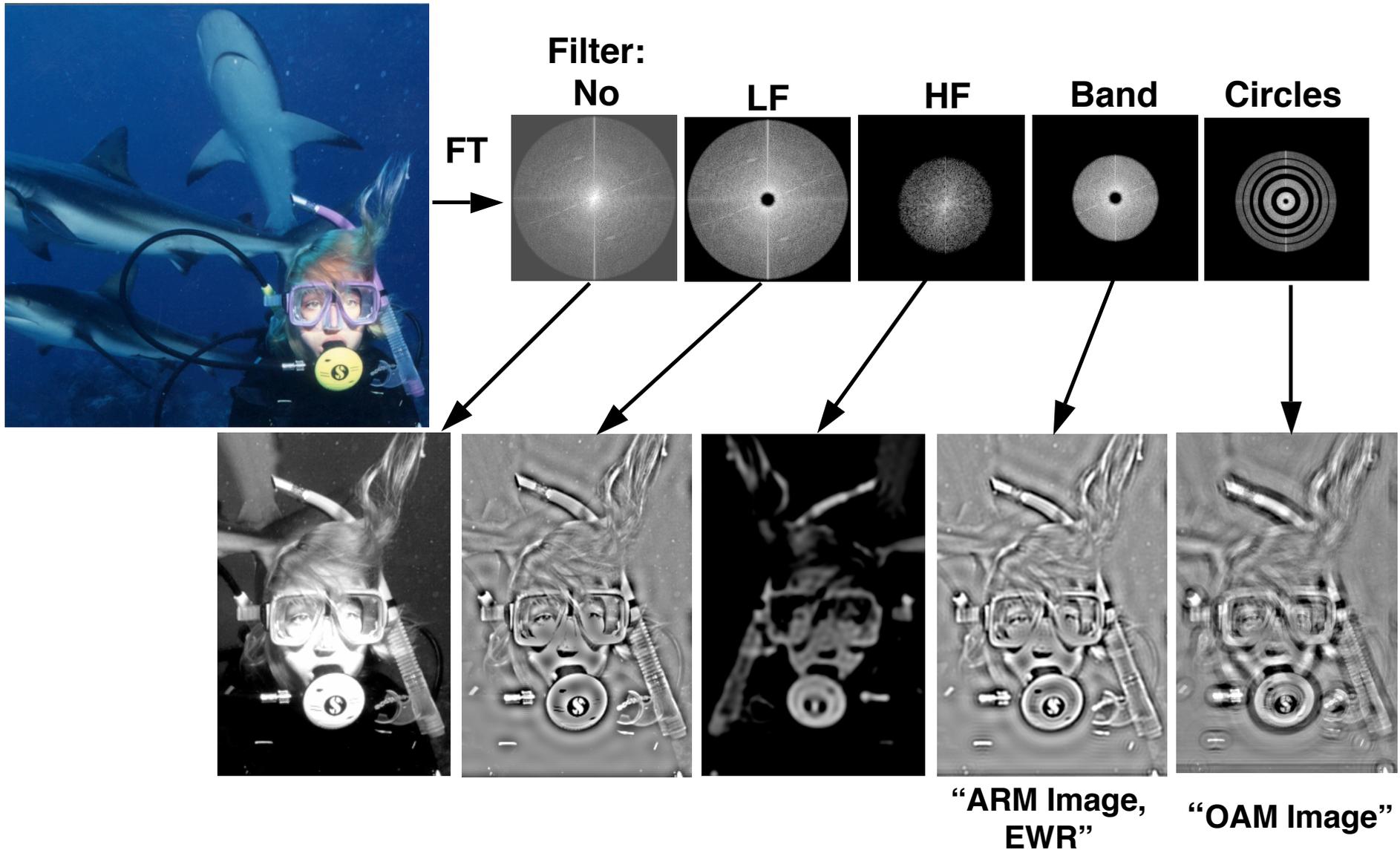
CTF oscillations mix amplitudes and phases

CTF oscillations increase image delocalization

- We gain resolution - at a huge price....

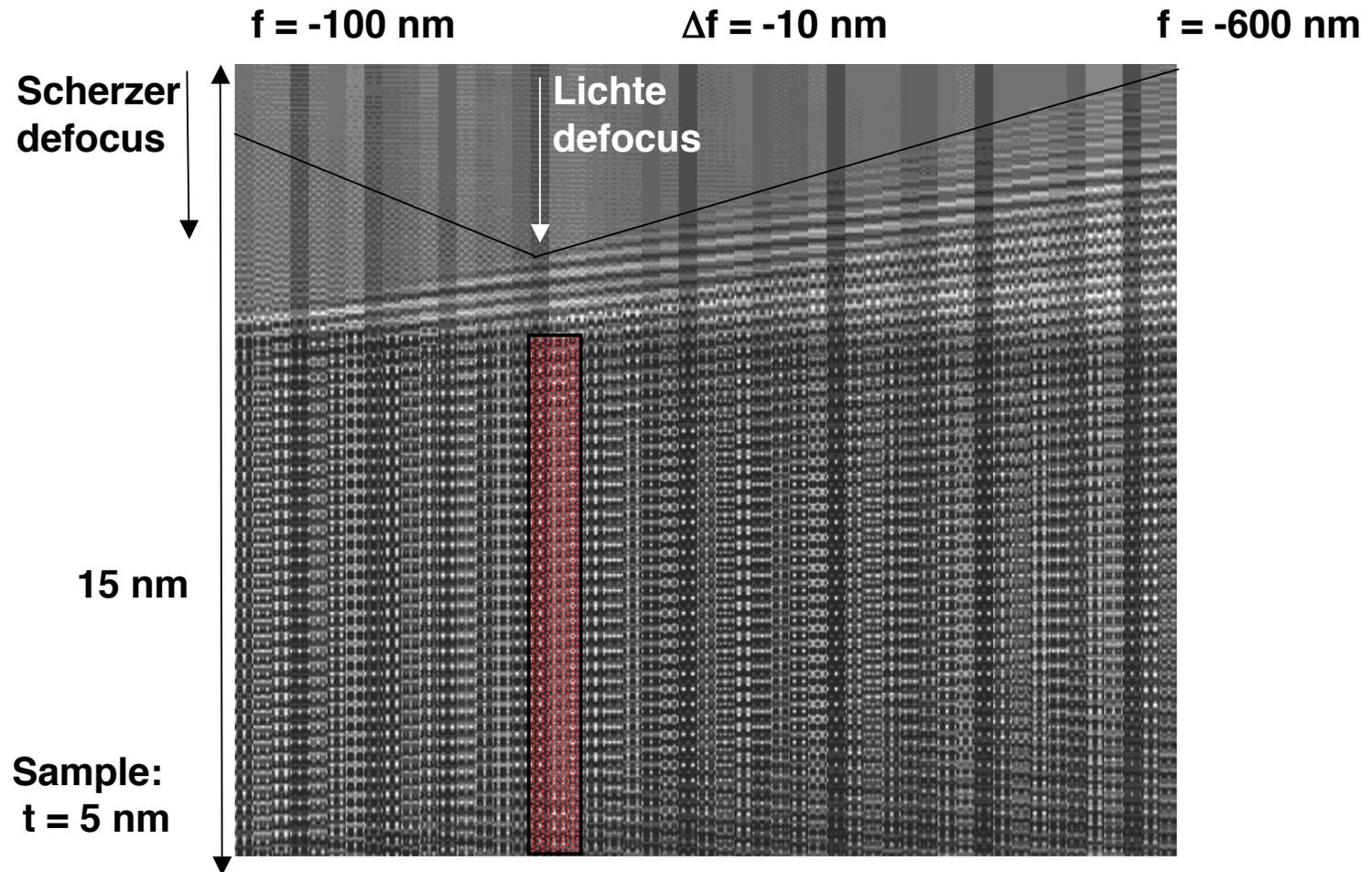
# Principle of HRTEM

## Image formation: delocalization



# Principle of HRTEM

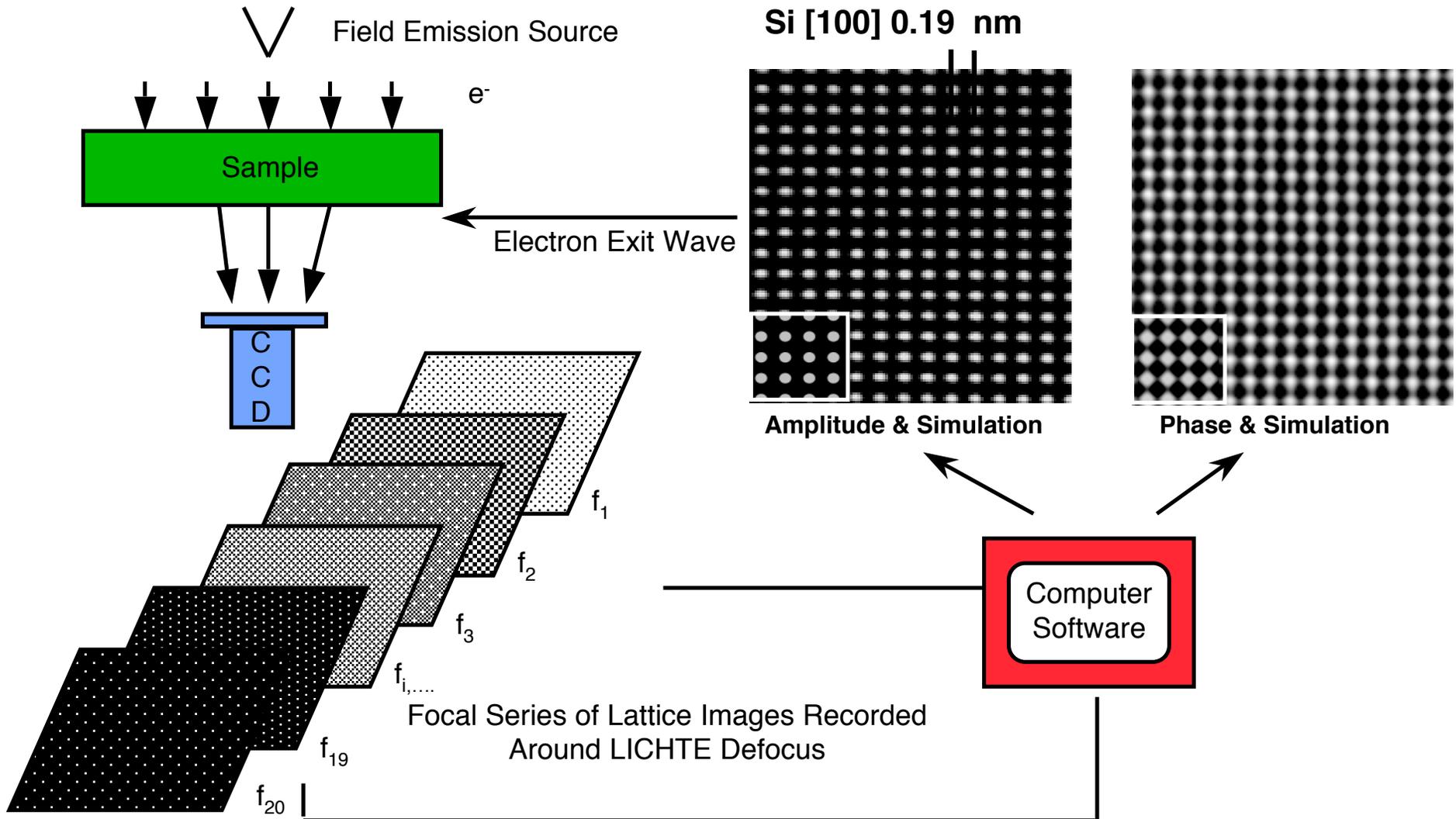
## Image formation: delocalization



- We must fill the gaps in Fourier space (Focus series)
- We can solve the phase problem simultaneously

# From Lattice Images to Electron Exit Waves

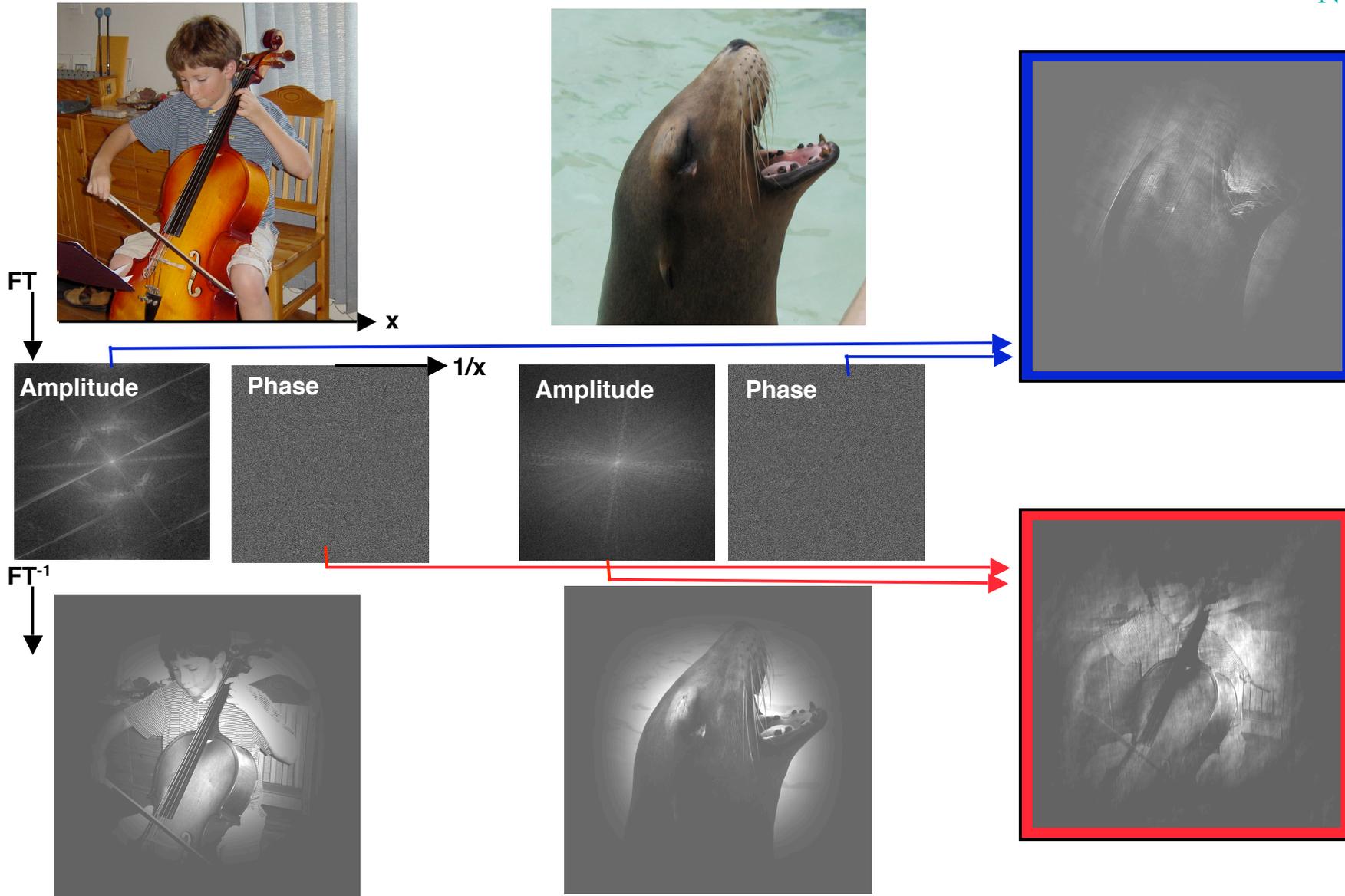
Solving the phase problem; removing aberrations & delocalization



W.M.J. Coene, A. Thust, Ultramicroscopy 64 (1996) 109, 211  
 F.R. Chen, R. Kilaas et al. 2005

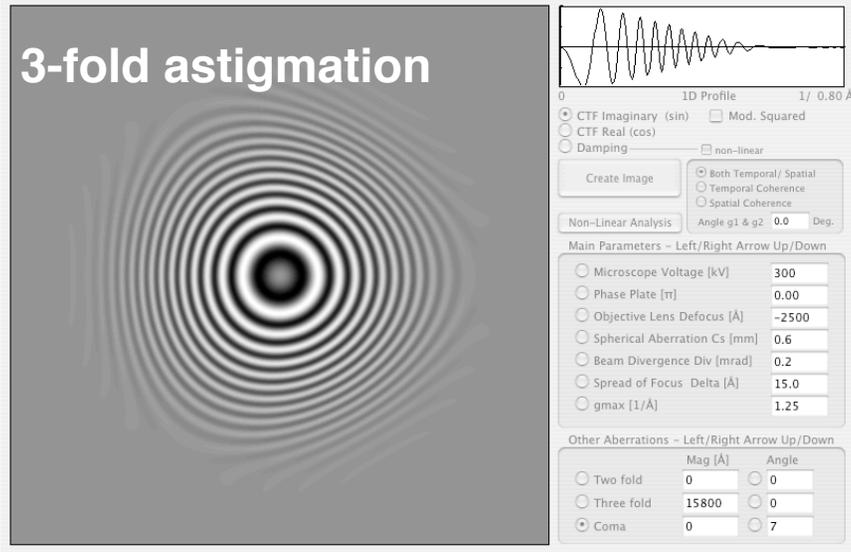
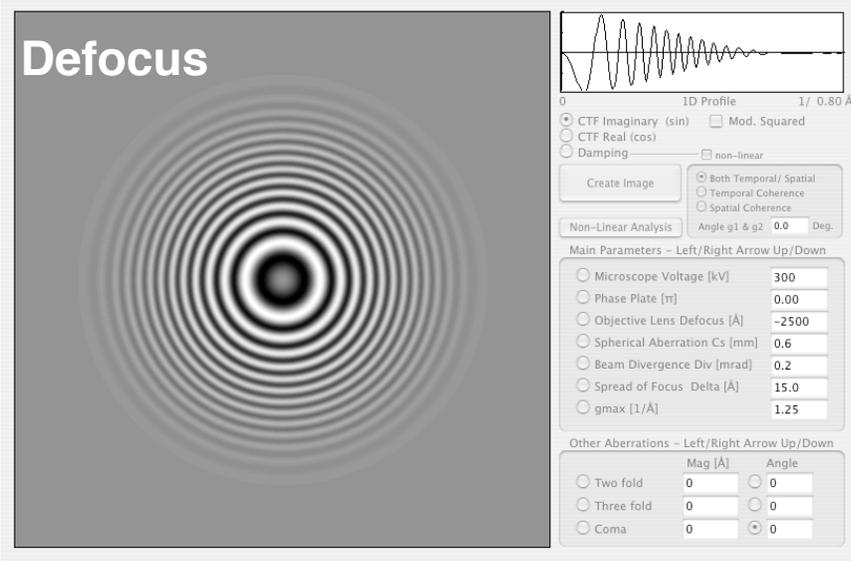
# On the Importance of Phases

Phases carry most of the information in an image

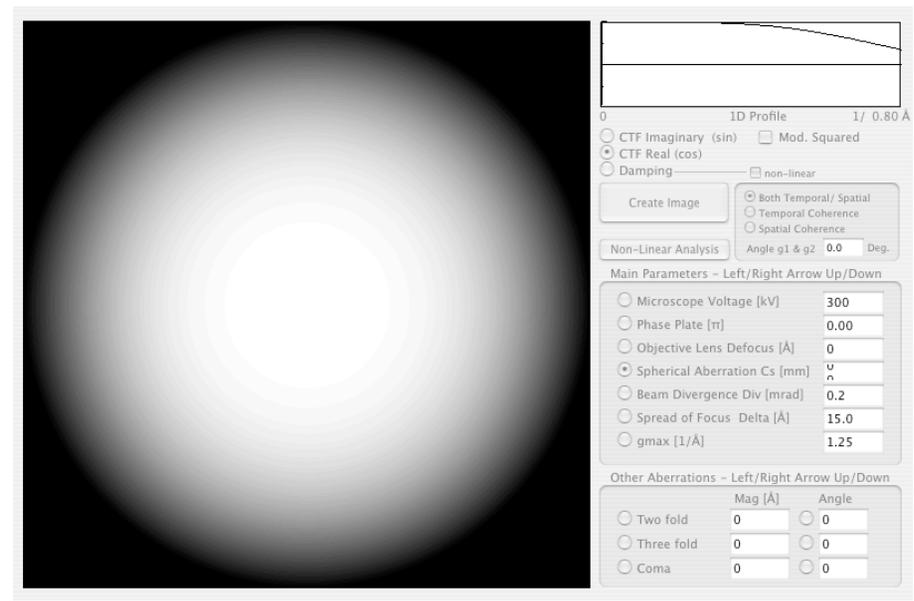


# Lens Aberrations

## TEM: Point spread function



### Cs corrected amplitude imaging



**In TEM aberration can be corrected by Software & hardware**



# Higher Order Lens Aberrations

$$\begin{aligned}
 E(\alpha, \bar{\alpha}) = & \operatorname{Re} \left\{ C_1 \alpha \bar{\alpha} + \frac{1}{2} A_1 \bar{\alpha}^2 \right. \\
 & + B_2 \alpha^2 \bar{\alpha} + \frac{1}{3} A_2 \bar{\alpha}^3 \\
 & + \frac{1}{2} C_3 \alpha^2 \bar{\alpha}^2 + S_3 \alpha^3 \bar{\alpha} + \frac{1}{4} A_3 \bar{\alpha}^4 \\
 & + B_4 \alpha^3 \bar{\alpha}^2 + D_4 \alpha^4 \bar{\alpha} + \frac{1}{5} A_4 \bar{\alpha}^5 \\
 & + \frac{1}{3} C_5 \alpha^3 \bar{\alpha}^3 + S_5 \alpha^4 \bar{\alpha}^2 + R_5 \alpha^5 \bar{\alpha} + \frac{1}{6} A_5 \bar{\alpha}^6 \\
 & + B_6 \alpha^4 \bar{\alpha}^3 + D_6 \alpha^5 \bar{\alpha}^2 + F_6 \alpha^6 \bar{\alpha} + \frac{1}{7} A_6 \bar{\alpha}^7 \\
 & \left. + \frac{1}{4} C_7 \alpha^4 \bar{\alpha}^4 + S_7 \alpha^5 \bar{\alpha}^3 + R_7 \alpha^6 \bar{\alpha}^2 + G_7 \alpha^7 \bar{\alpha} + \frac{1}{8} A_7 \bar{\alpha}^8 \right\}.
 \end{aligned}$$

name	symbols						
defocus	$C_1$						
n-th order spherical aberration			$C_3$		$C_5$		$C_7$
n-th order axial coma		$B_2$		$B_4$		$B_6$	
n-th order star aberration			$S_3$		$S_5$		$S_7$
n-th order three-lobe aberration				$D_4$		$D_6$	
n-th order rosette aberration					$R_5$		$R_7$
n-th order pentacle aberration						$F_6$	
n-th order chaplet aberration							$G_7$
v-fold astigmatism	$A_1$	$A_2$	$A_3$	$A_4$	$A_5$	$A_6$	$A_7$

M. Haider

- **There are axial (listed) and off-axial aberrations**
- **Often only  $C_1$  ( $\Delta f$ ) and  $C_3$  ( $C_s$ ) are considered**

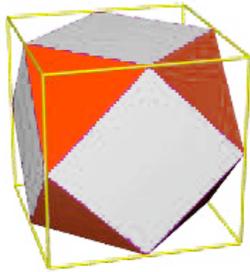
$$\chi(g) = 1/2 \Delta f \lambda g^2 + 1/4 C_s \lambda^3 g^4$$
- **OAM:  $A_1, A_2, A_3, B_2, (C_1 \text{ and } C_3 \text{ by EWR})$**
- **Higher order aberrations become important if the resolution increases (TEAM-Project --- 0.05 nm)**
- **It is challenging to correct all aberration by software only**



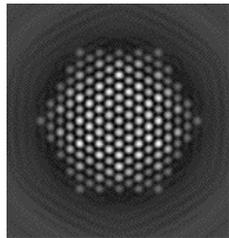
# Cs Corrected Microscopy

## Advantages

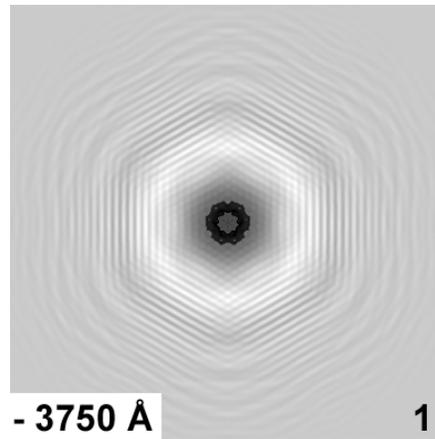
NCEM —



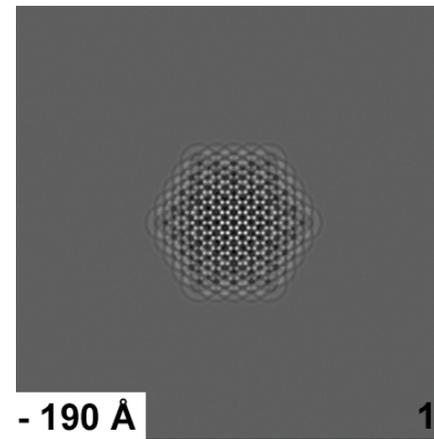
cuboctahedron



Exit wave



Focal series: CM300 (150 kV)

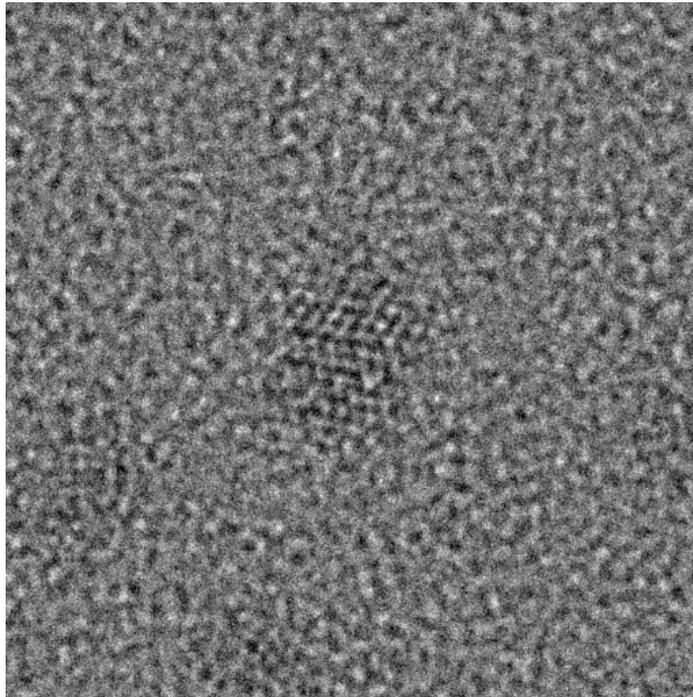


Cs corrected TEAM (150 kV)

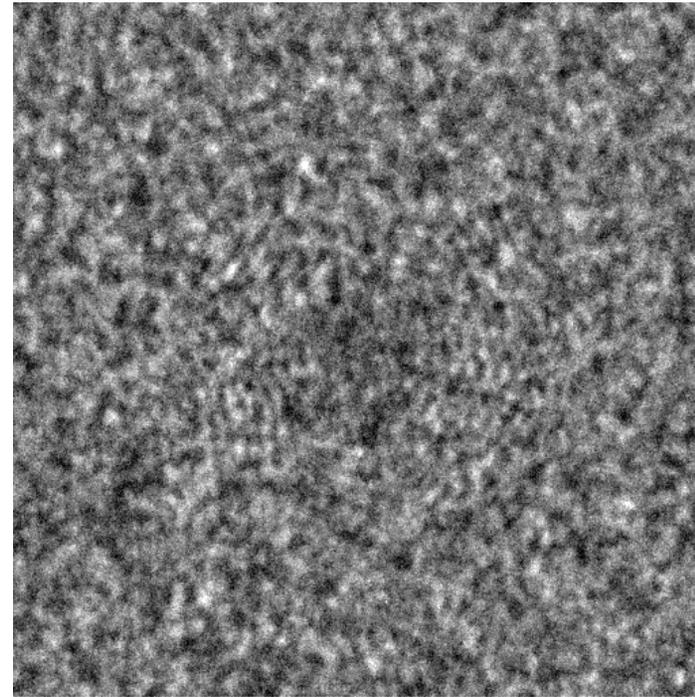
- **Image delocalization of small particles can wipe out lattice fringes**
- **Cs correction reduces the effect largely**
- **It will always be beneficial to do EWR in a Cs corrected microscope**
- **Access to waves**
- **What happens in the presence of noise?**

# Software correction

## Experimental limitations



FePt on carbon, Scherzer defocus

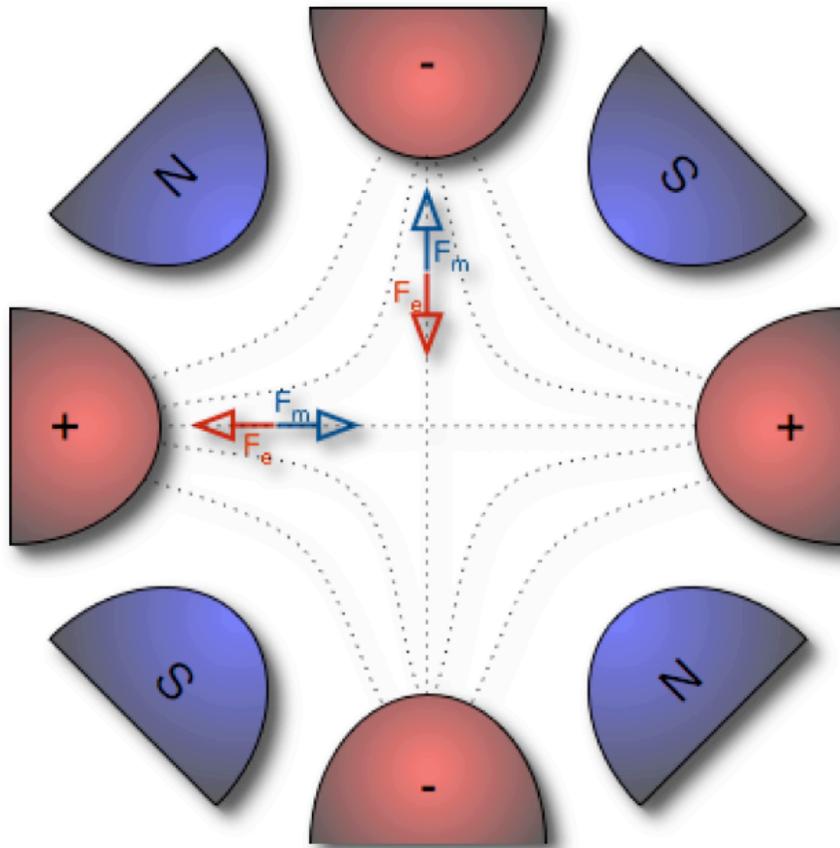


$\Delta f \sim$  Lichte defocus (-260 nm)

- **Experiment:**  
**Image delocalization can be a significant limitation for the reconstruction of small particles on a carbon film**

# Hardware Correction: Wien Filter

Crossed electric + magnetic quadrupoles



Use magnetic and electric fields to deflect electrons in opposite directions.

$$\text{for } v_0 = E / B, \quad \mathbf{F} = 0$$

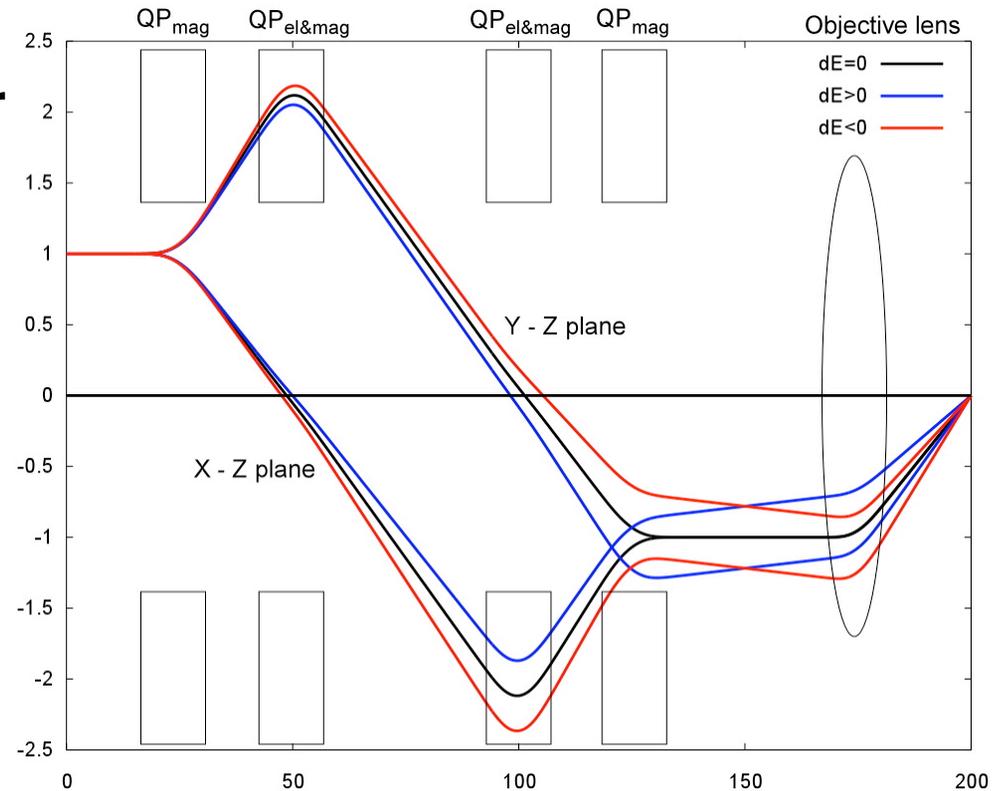
Electrons with  $v_0$  will not be affected. All other “colors” will be distorted.

$$\mathbf{F} = q\{\mathbf{E} + (\mathbf{v} \times \mathbf{B})\}$$

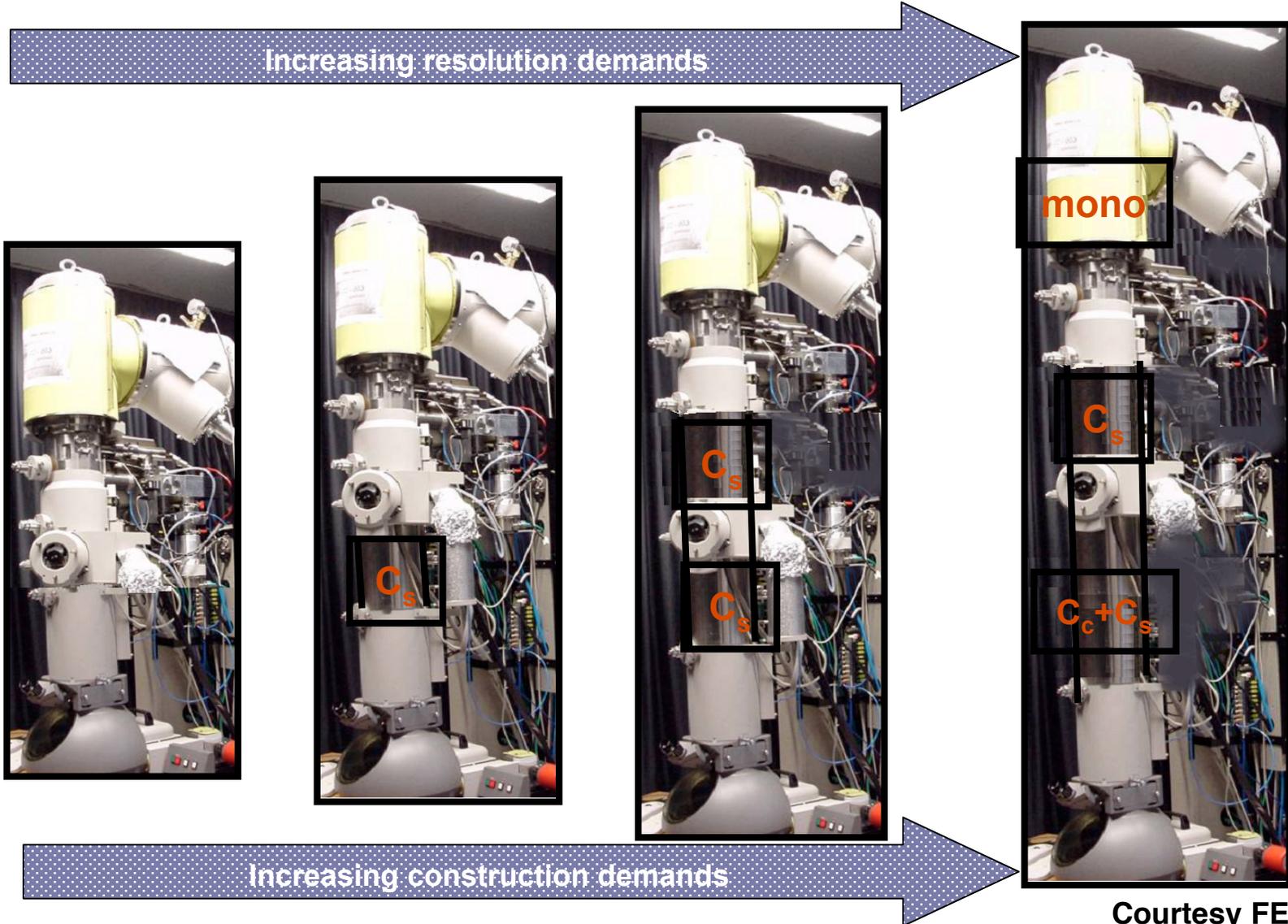
# Approach to Aberration Correction

Rose - CEOS

- **Superaplanator - Cs/Cc corrector**
  - 2 Wien filters
  - 3 octopole elements
  - Use of symmetry to cancel aberrations.
- **Cc/Cs correctors are tall**
  - 0.8 - 1 m

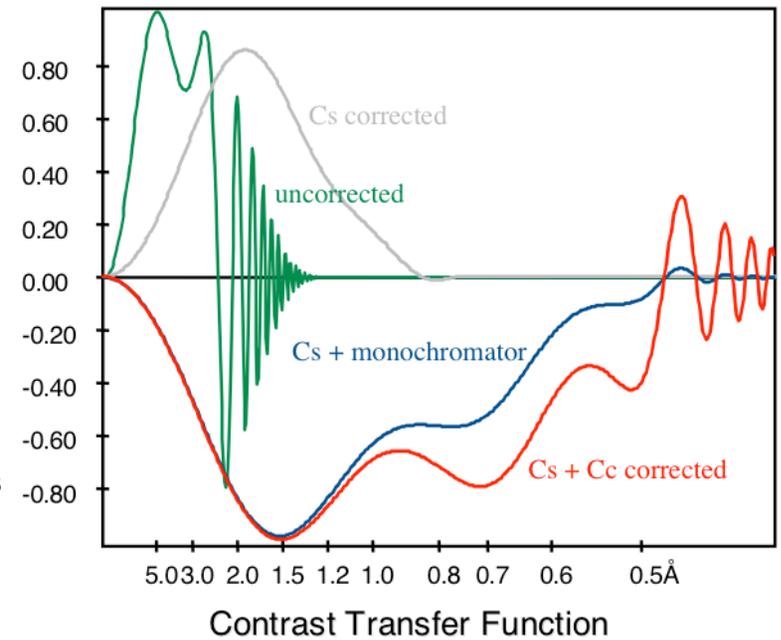
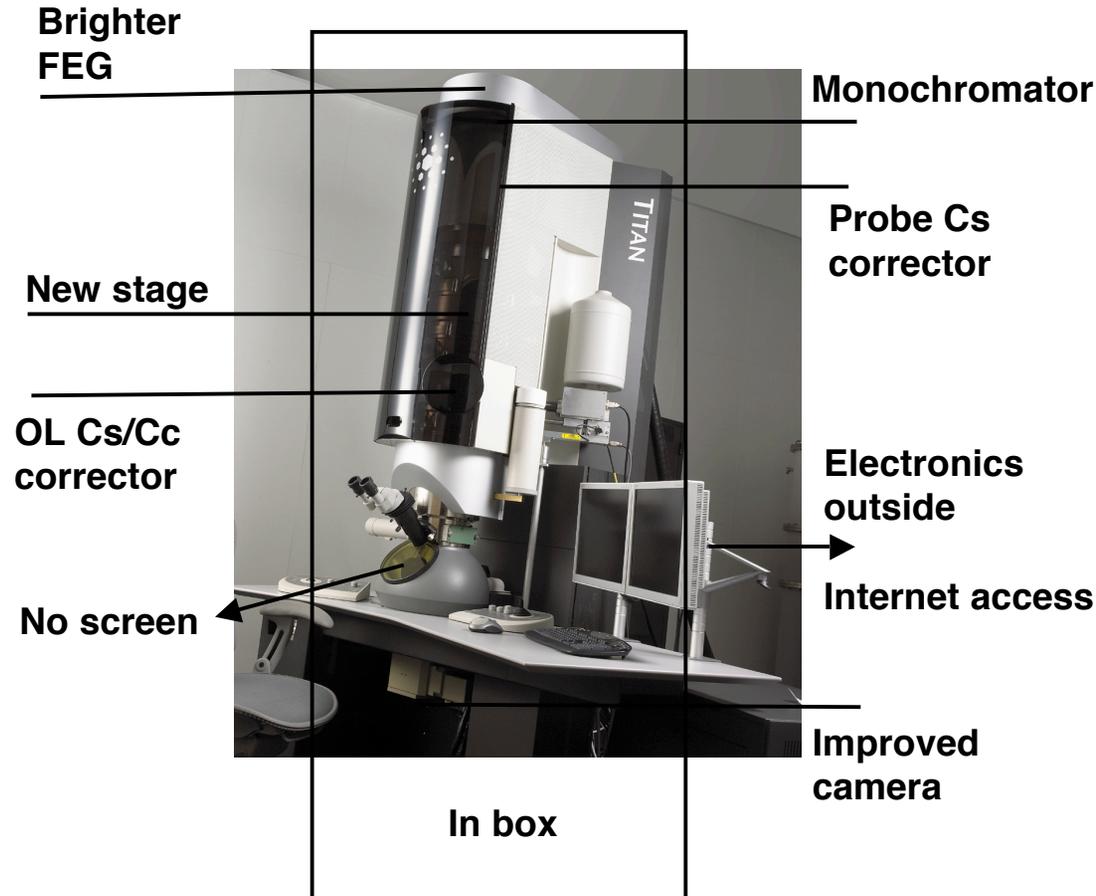


# Consequences for Design



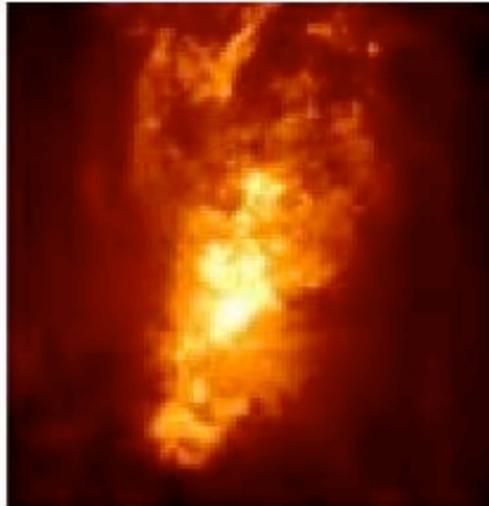
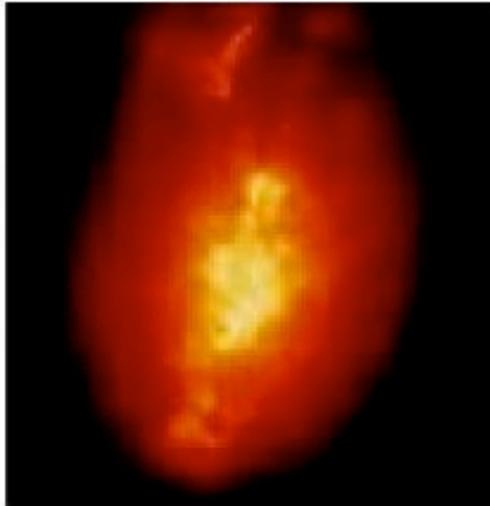
Courtesy FEI Company

# From Titan to TEAM

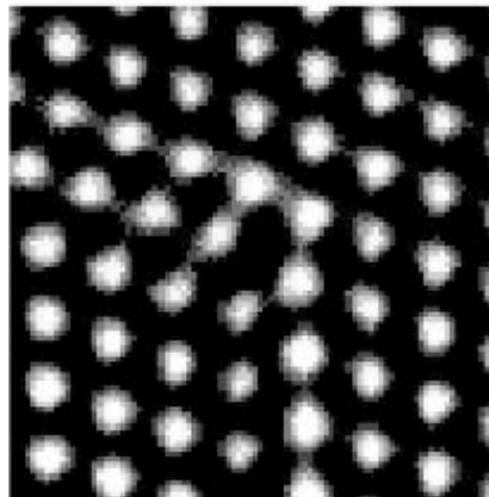
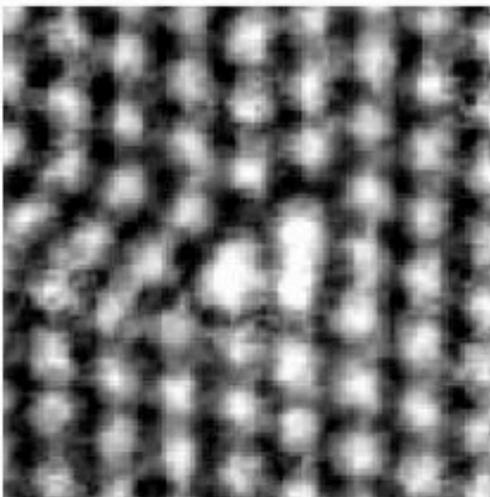


Numbers seem small but it is a long way from 0.8Å to 0.5Å in STEM/TEM

# TEAM Gets Us “Sharper” Images



- Hubble telescope before and after correction of spherical aberration (1993)



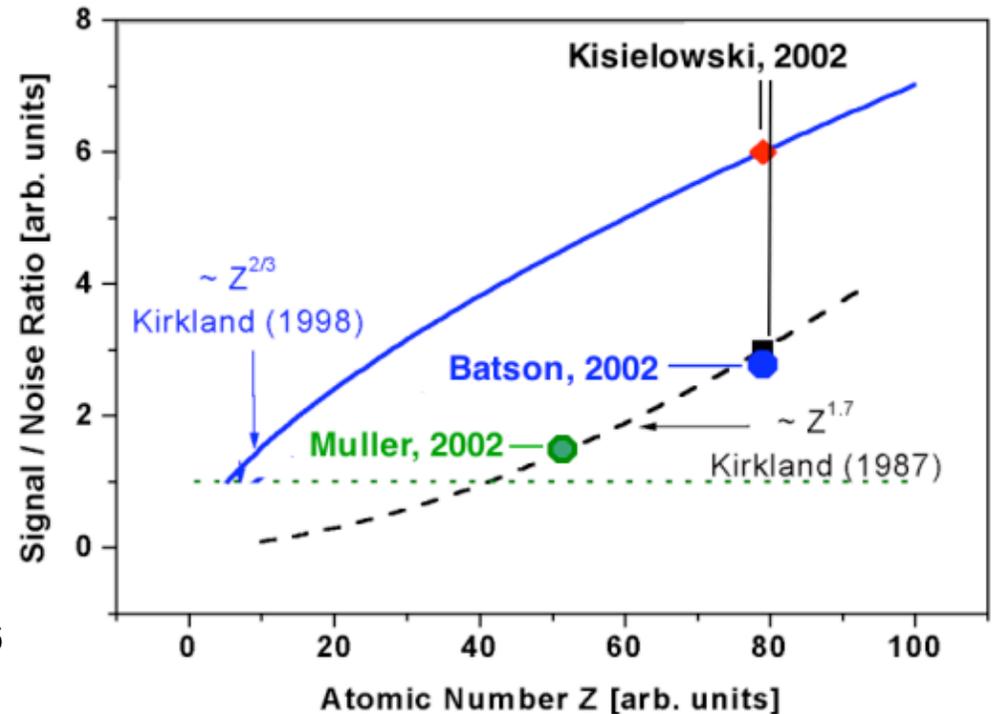
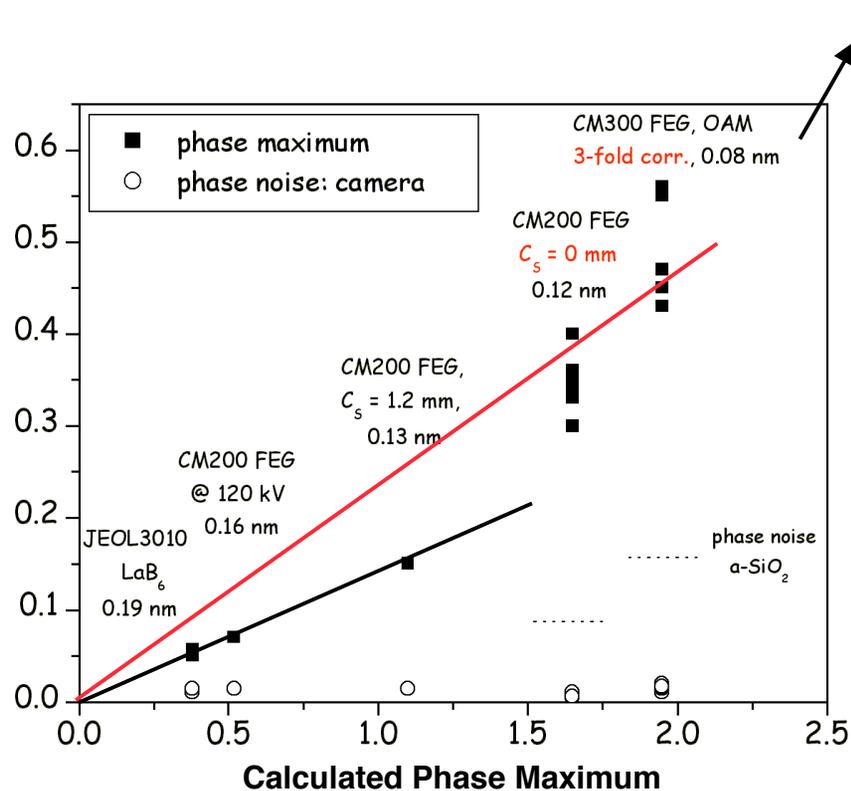
- Atomic resolution images of dislocation core in Au before and after correction of spherical Aberration

- How much is the improvement?

Penisson, Berkeley ARM

Kisielowski, Juelich, 2002

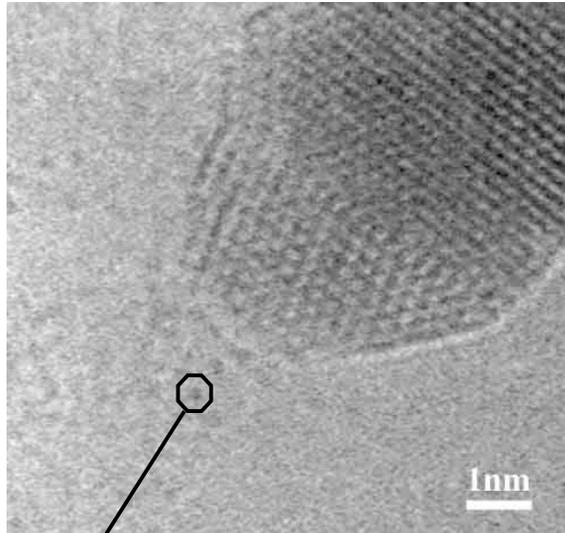
# More Important than Resolution: Sensitivity



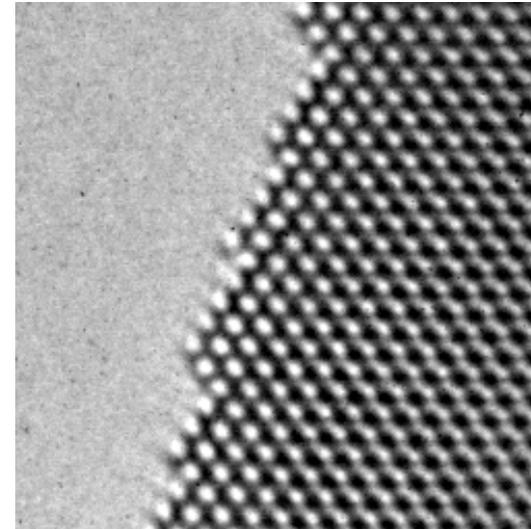
- Technological progress with EM has boosted sensitivity  
FEG, aberration correction,...
- Single atoms of most elements from the Periodic Table can be detected
- TEAM Project will provide the “ultimate” resolution / sensitivity

# Single Atom Sensitivity: Current abilities

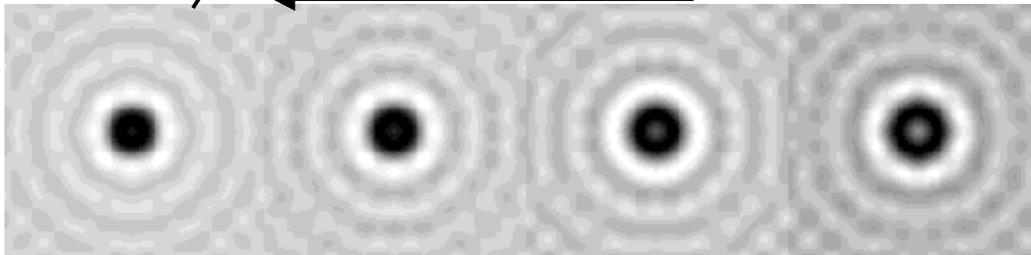
OAM: 30 frames/sec



Cs Correction: 1 frame/sec



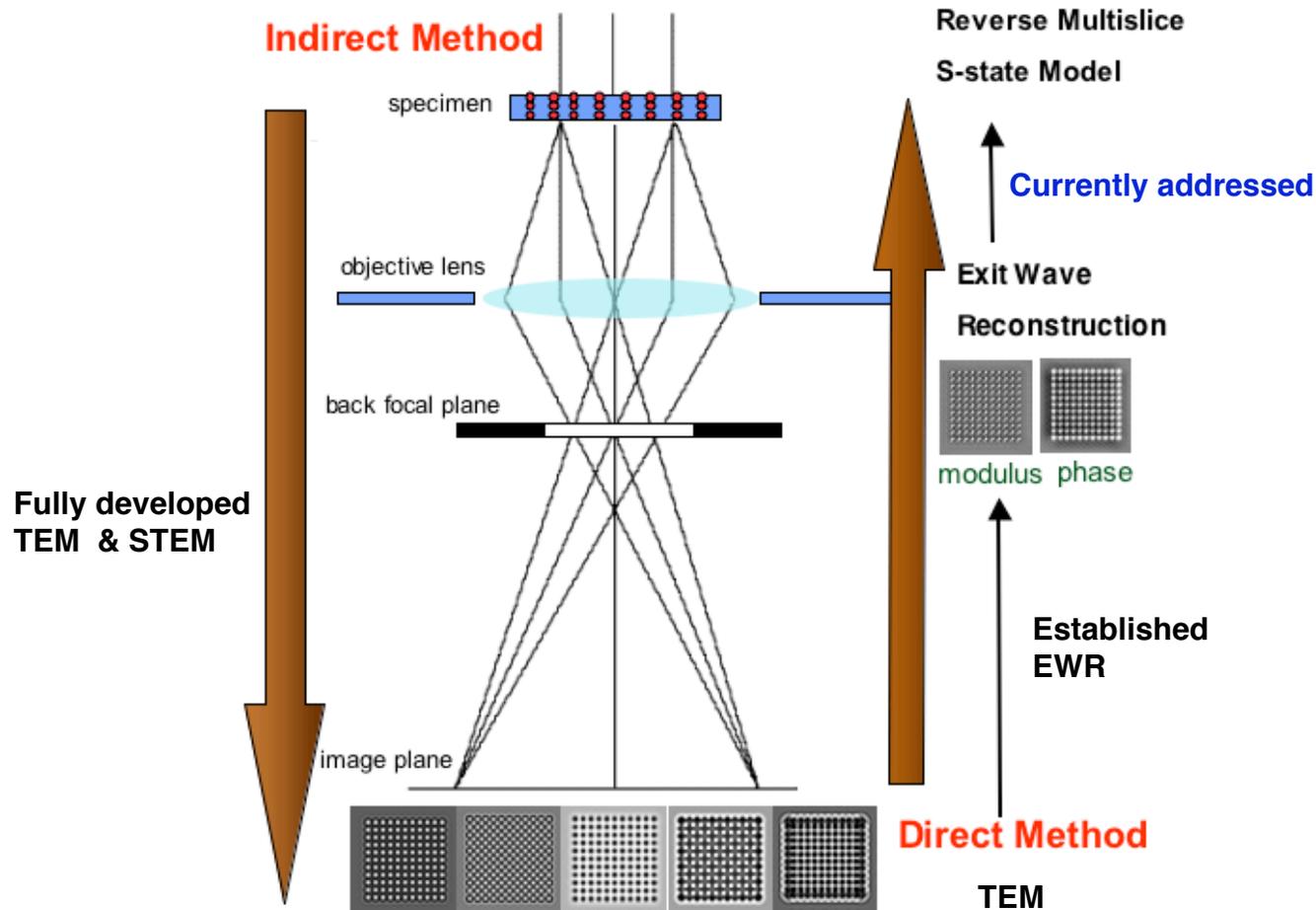
$\Delta f$  (Simulation)



Intensity reflects atom position

**Brighter gun & better cameras allow for better real time experiments with large S/N ratio**

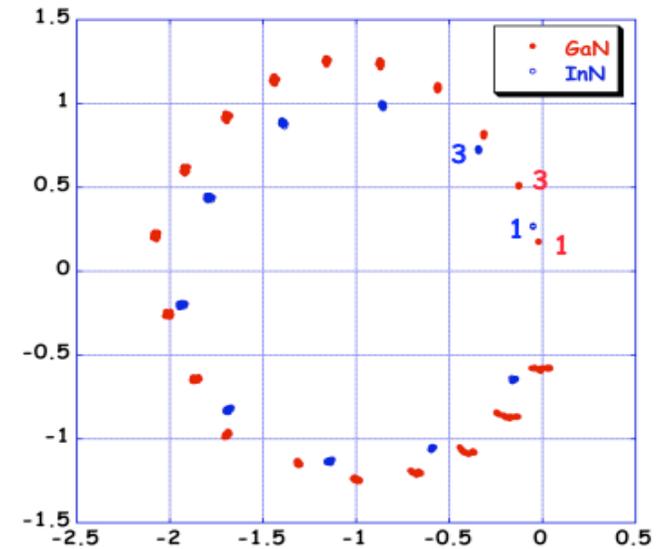
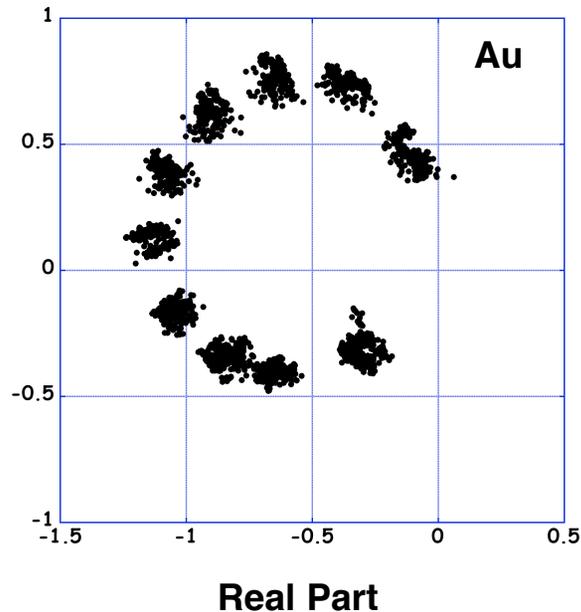
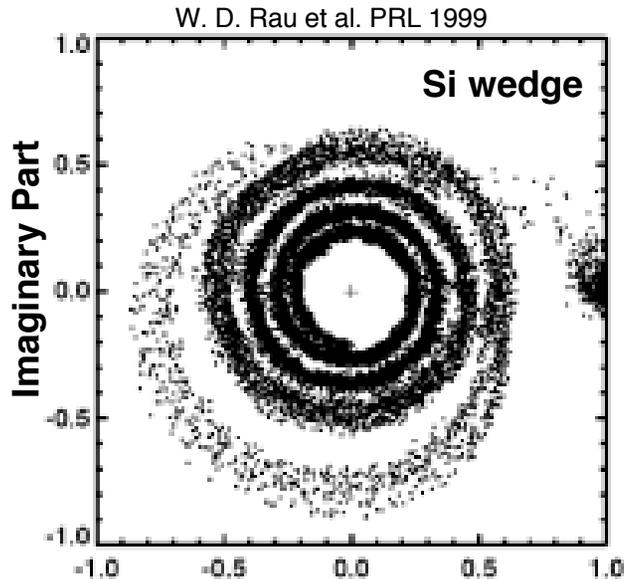
# Experiments Must be Guided and Verified by Simulations



- Past: Utilization of indirect methods (lack of uniqueness)
- Today: Development of direct methods
  - Progress with theory has enabled a reliable reconstruction of electron exit waves
  - Structure reconstruction from exit ways becomes feasible

# Interpretation of Waves

## Amplitude - phase diagrams (Argand plot)



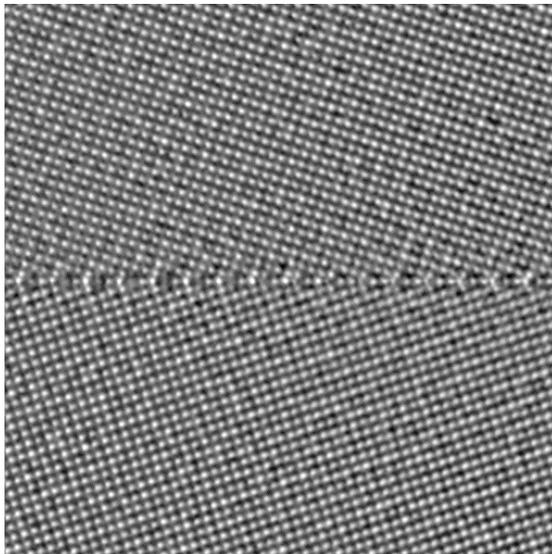
- A wave is entirely characterized by amplitude-phase diagrams
- In high performance microscopes Argand plots can be discrete
- Chemical composition can be revealed with single atom sensitivity
- Argand plots depict the full information - there is nothing more



# Intensity Interpretation

## Current abilities; example Al:Cu

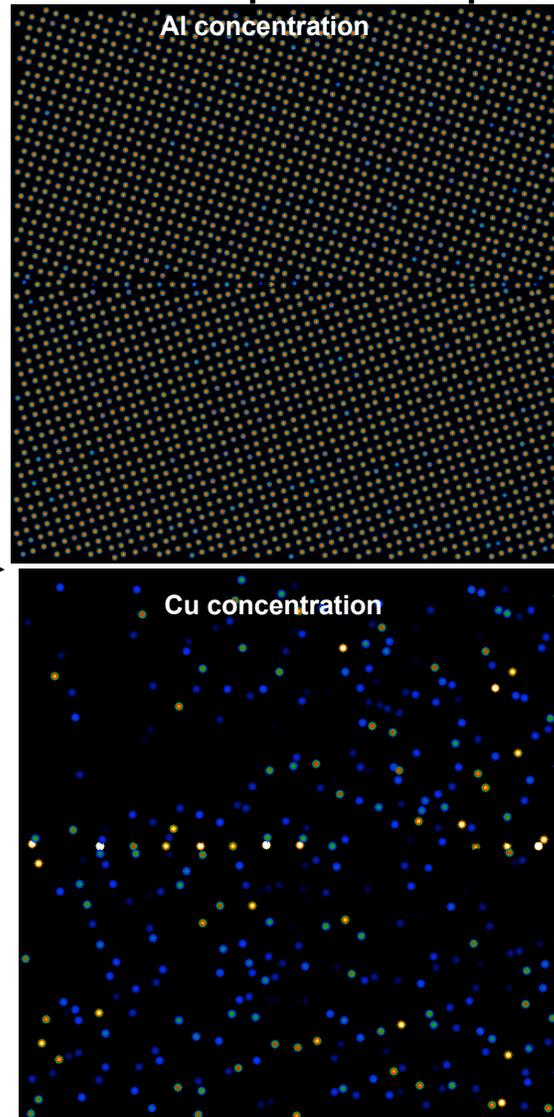
Exit wave (phase)



Al:Cu

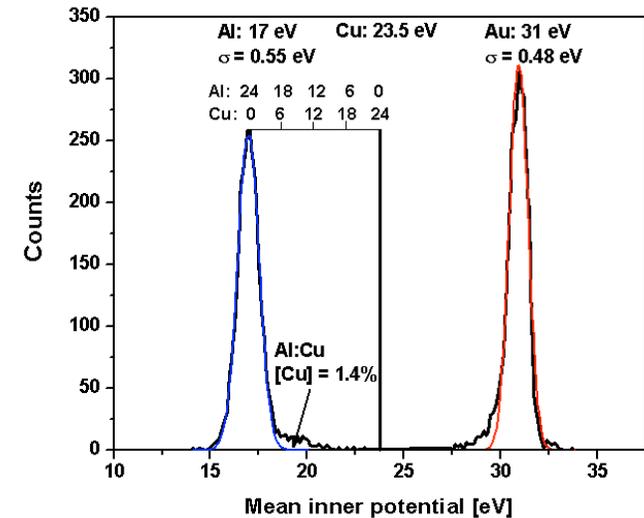
F.R. Chen, C. Kisielowski, work in progress

Retrieved potential maps



### Requirements:

- Recover waves
- Inversion of dynamic scattering

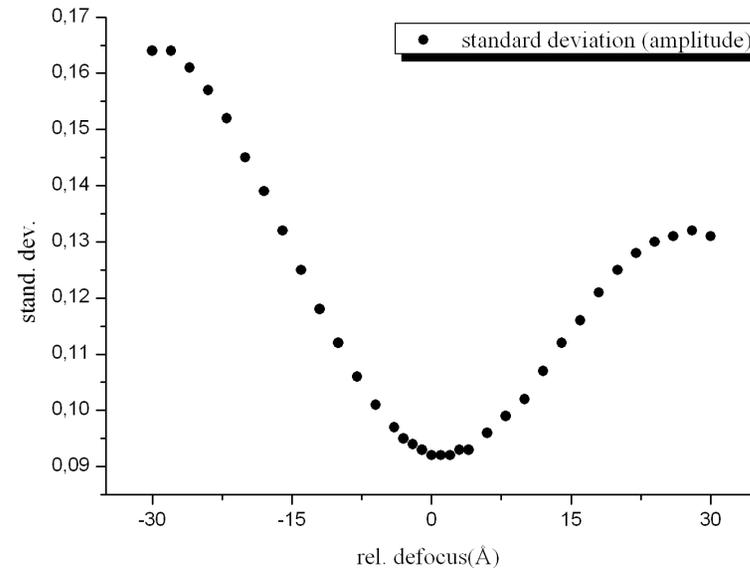
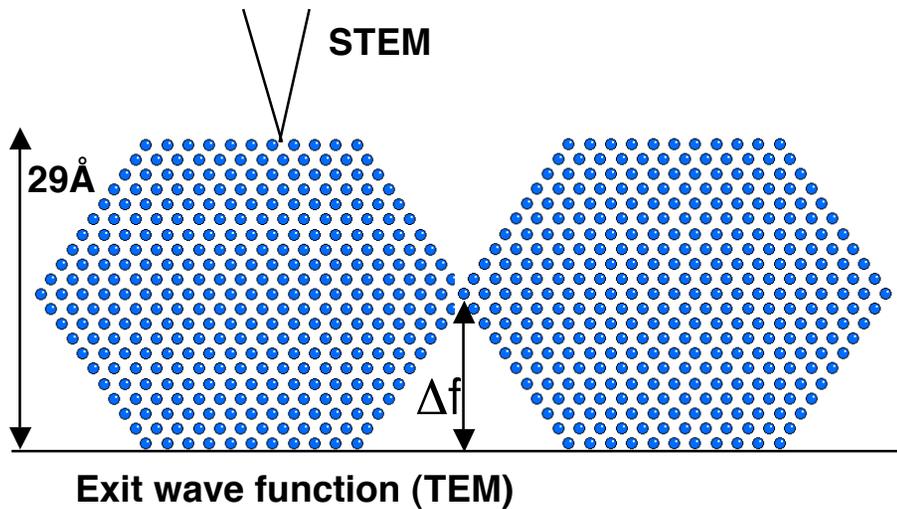


### Limits

- Given by energy spread
- Recognizing a substitution of 1 Cu in column of 24 Al:  $\sigma = 0.27$  eV

# Limitations

Sample geometry, sample preparation & instrument stabilities



- The shape of nanoparticles can introduce local defocus changes
- Absolute amplitude/phase values noticeably over 2 Å
- Demands on stability are extremely high



# Exploiting Sensitivity: Discrete Tomography

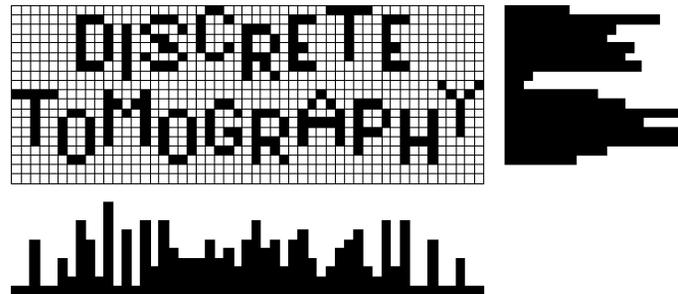
NCEM —

WORKSHOP ON

## DISCRETE TOMOGRAPHY AND ITS APPLICATIONS

June 13-15, 2005

[New York City](#)



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### Organizers

[Gabor T. Herman](#)

[Attila Kuba](#)

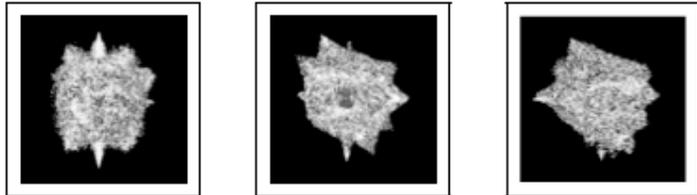
- Workshops on DT are held regularly ( ~ 100 participants)
- Discrete (binary) tomography is a general approach (biology, medicine, solids,...)
- It is very suited to be applied to crystalline solids since atoms are discrete objects

## Binary Tomography

*K.J. Batenburg / Electronic Notes in Discrete Mathematics 20 (2005) 247–261*



Original volume of size  $128 \times 128 \times 128$ .



Reconstruction from two projections:  $(1, 0, 0)$  and  $(0, 1, 0)$ .



Reconstruction from three projections:  $(1, 0, 0)$ ,  $(0, 1, 0)$  and  $(0, 0, 1)$ .



Reconstruction from four projections:  $(1, 0, 0)$ ,  $(0, 1, 0)$ ,  $(0, 0, 1)$  and  $(1, 1, 0)$ .

259 •Utilizing constraints reduces the number of projections

•Binary information is a powerful constraint

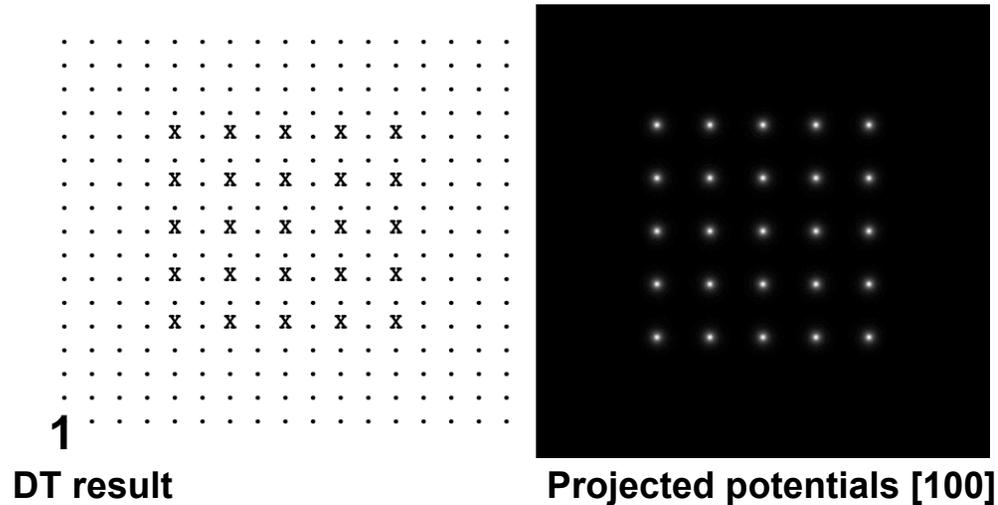
•Complex structures can be reconstructed

•Atomic resolution (S)TEM may give binary information  
Materials are made from single atoms  
Constraint: atomicity

•Atomicity is NO limiting constraint for (S)TEM

•How can the number of atoms in columns be counted?  
Utilize imaging along zone axis  
(dynamic scattering)

# Discrete Tomography Simulation

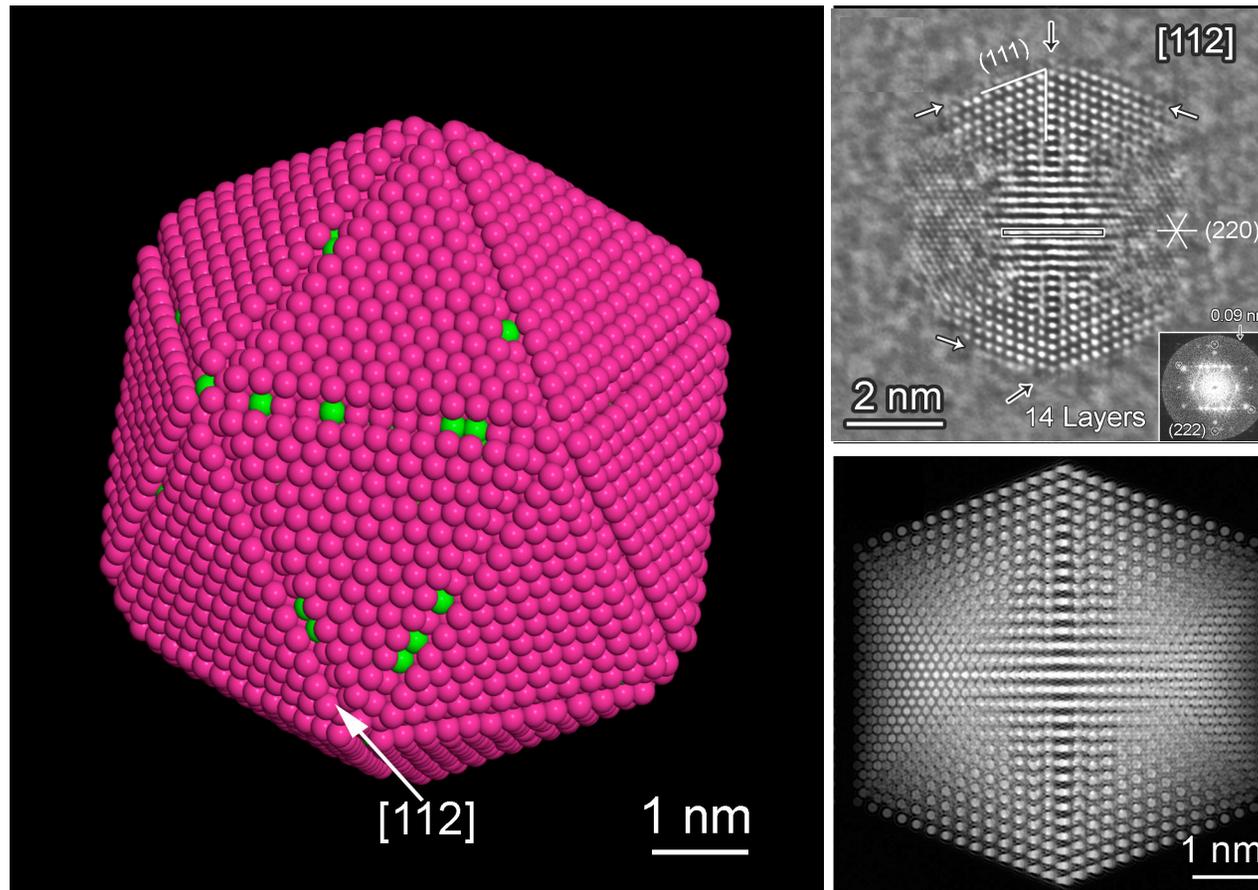


#projections	3				5				6			
orientations	[001], [110], [110]				[001], [110], [110], [011], [111]				[001], [110], [110], [011], [111], [111]			
#count errors	10	20	40	80	10	20	40	80	10	20	40	80
% perfect	36	3	0	0	100	97	61	6	100	99	80	39
avg. #atom errors	1.28	4.51	11.92	27.13	0	0.03	0.57	3.73	0	0.01	0.23	1.18

309 atoms

- Reconstruction agrees with input structure: 3D can be done
- Number of projections determines error bars
- Experimentally, availability of sample holders is limiting

# A Glimpse into the Future: Tomography with atomic resolution



- FePt icosahedra with core-shell structure (Nanomagnets with a catalytic shell)
- Strain relaxation by element redistribution and site-specific atom loss

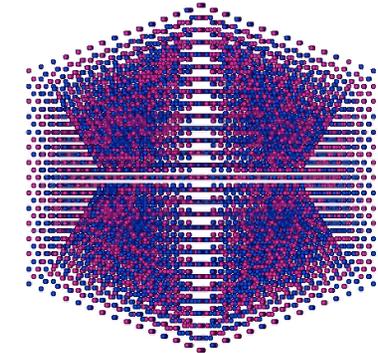
R. Wang et al. 2005, Nature Mat., 2005, in review

• **Aberration corrected EM enables novel science**

TEAM will meet the Feynman Challenge

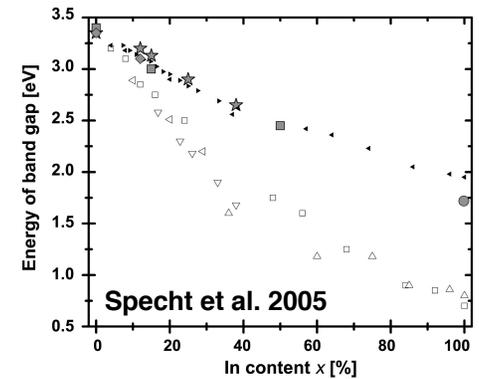
TEAM enables experiment design to meet scientific needs

- Choose the best of STEM/TEM in one instrument
- Do it in 3D

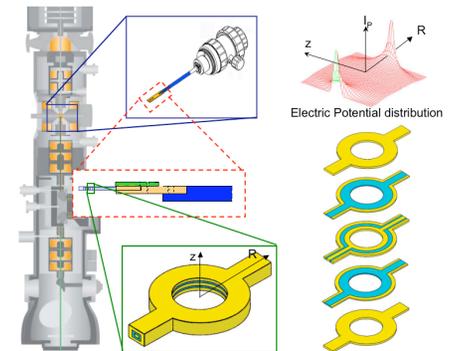


• **Many Developments will largely benefit from TEAM Project**

- Monochromator
- In-situ experiments (A. Minor)
- Soft materials (phase plate)
- Cameras (P. Denes)
- Stages (A. Schmid)
- .....



• **There is plenty of room for innovation**



W.K Hsieh et al. work in progress